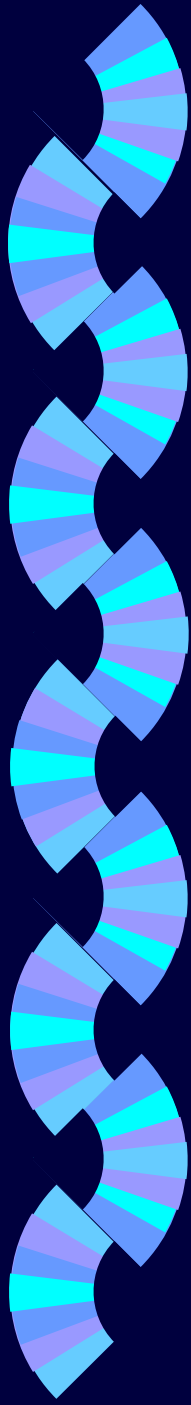


Nuclear reaction analysis and narrow profiling

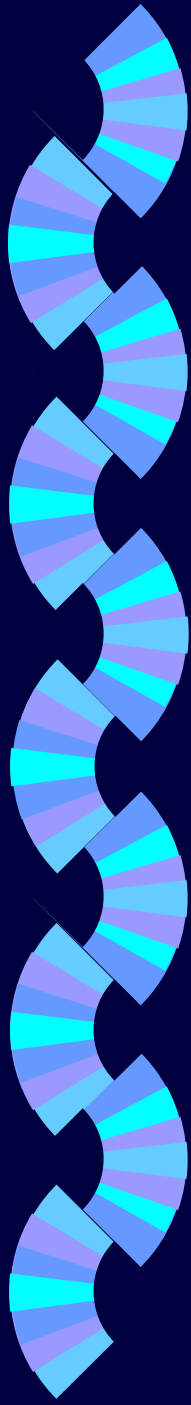
Fernanda Chiarello Stedile

fernanda.stedile@ufrgs.br



Topics

- Introduction to Si and SiC
- Principles of Isotopic Tracing
- Principles of Nuclear Reaction Analyses: NRA and NRP
- Results on thermal oxidations of Si and SiC



Advantages of Si

- Oxide film thermally grown (SiO_2)
Excellent electrical and thermodynamic characteristics
- SiO_2/Si interface
Low density of electrically active states

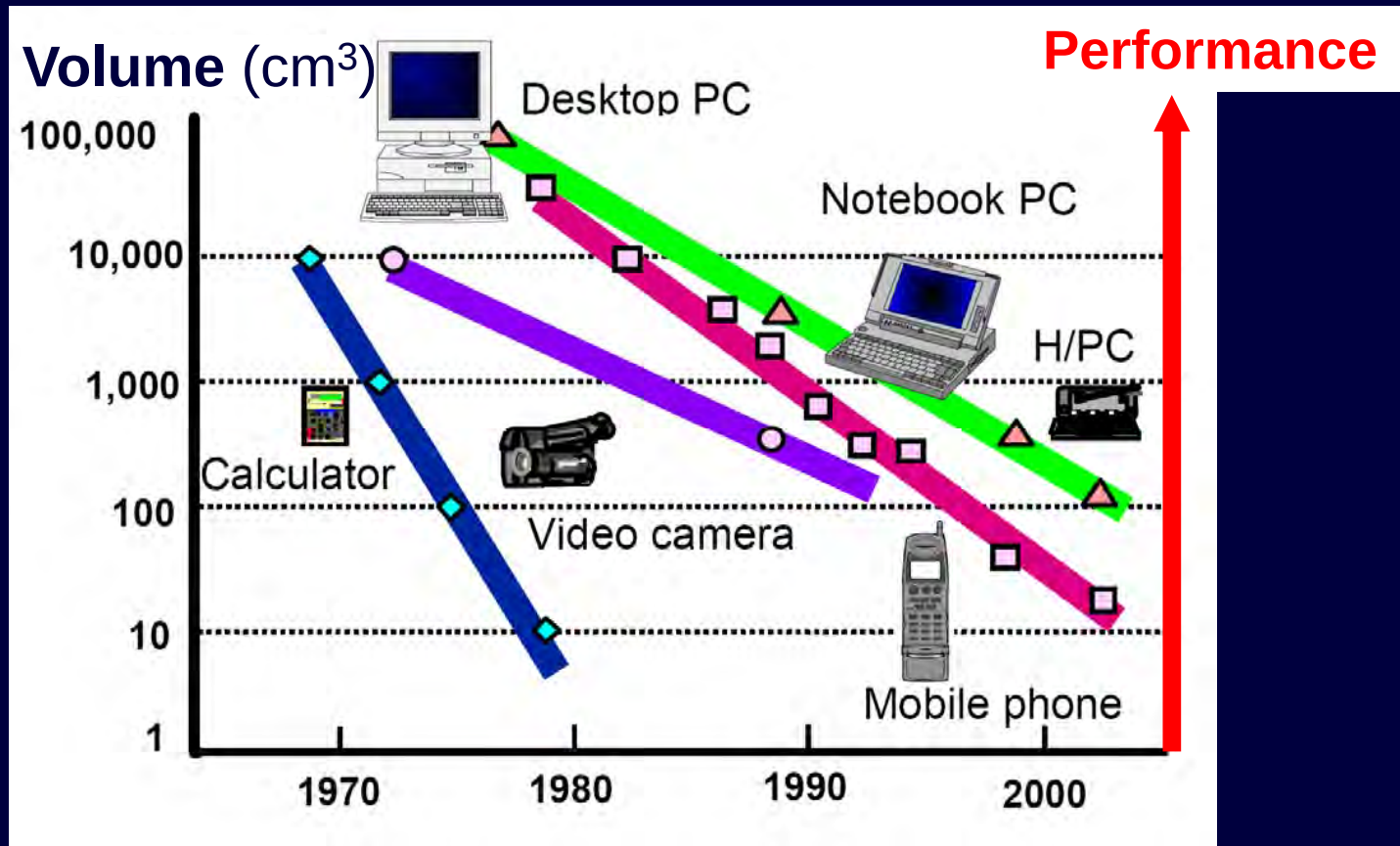
1993



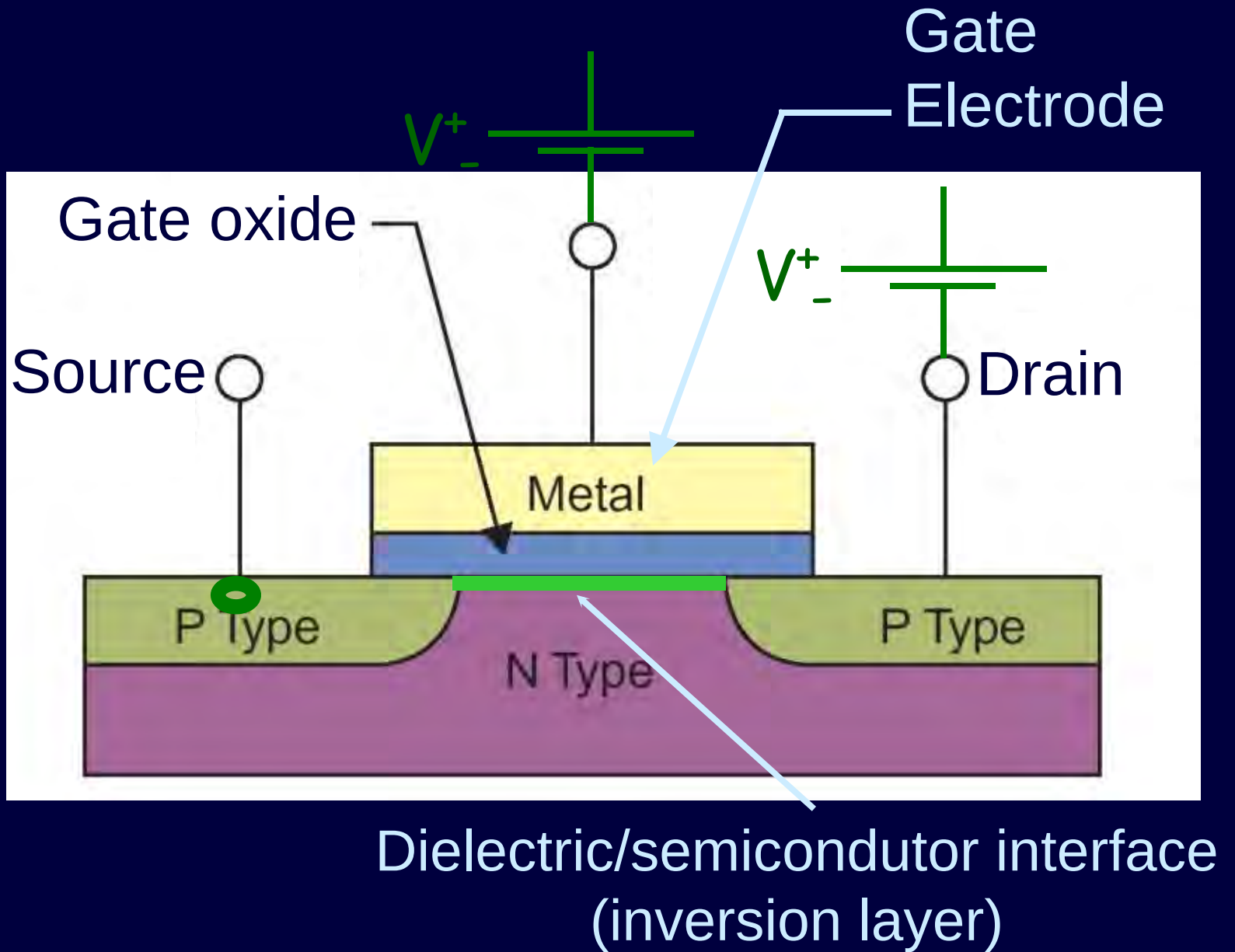
2013



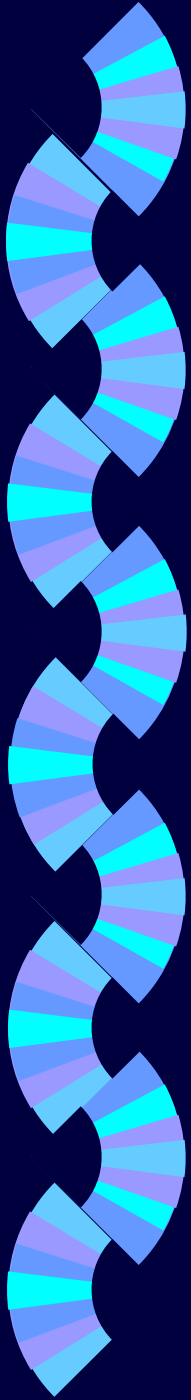
Silicon: the most widely used semiconductor in the microelectronic industry



MOSFET

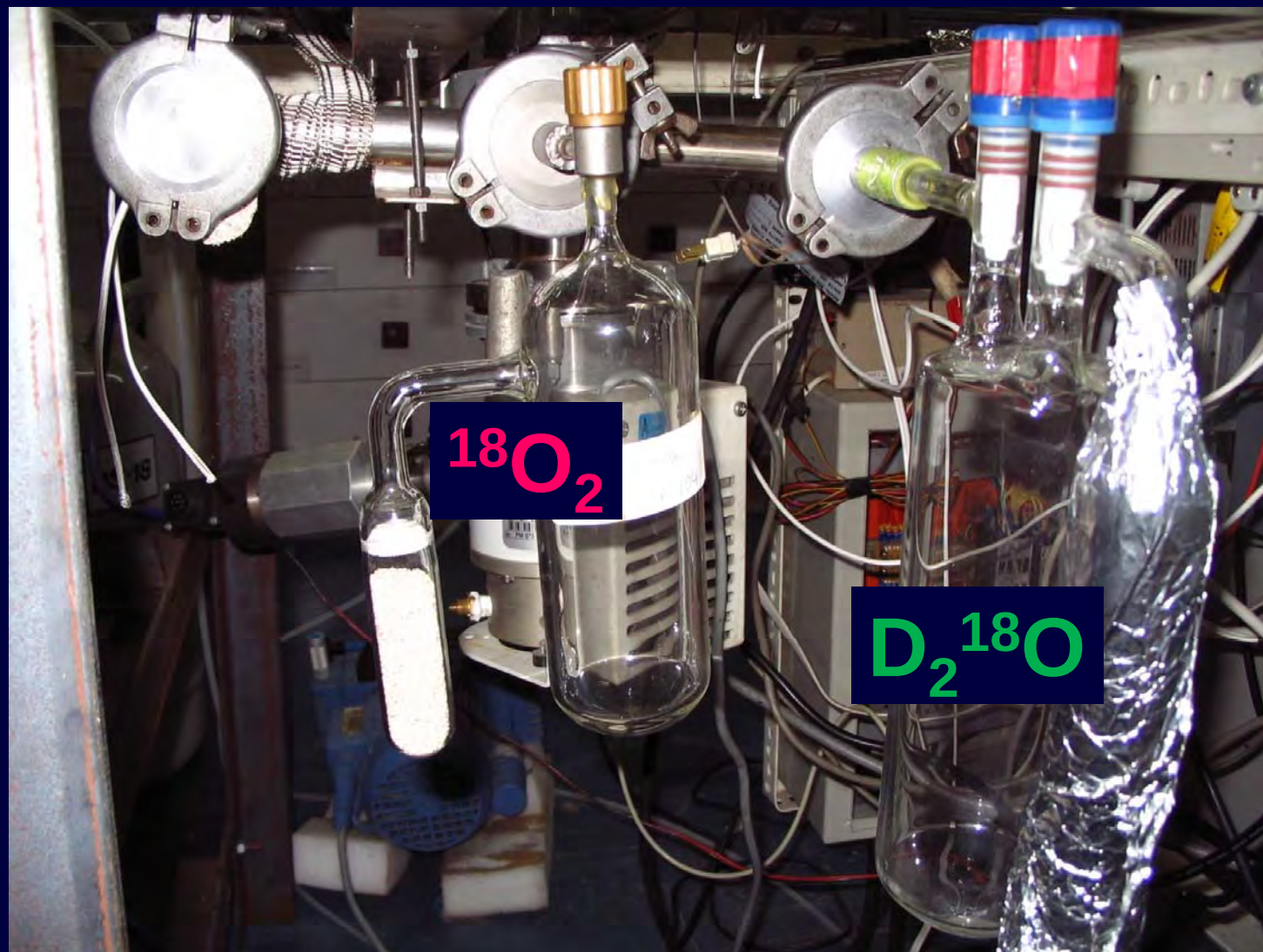


From chip to transistor - Intel



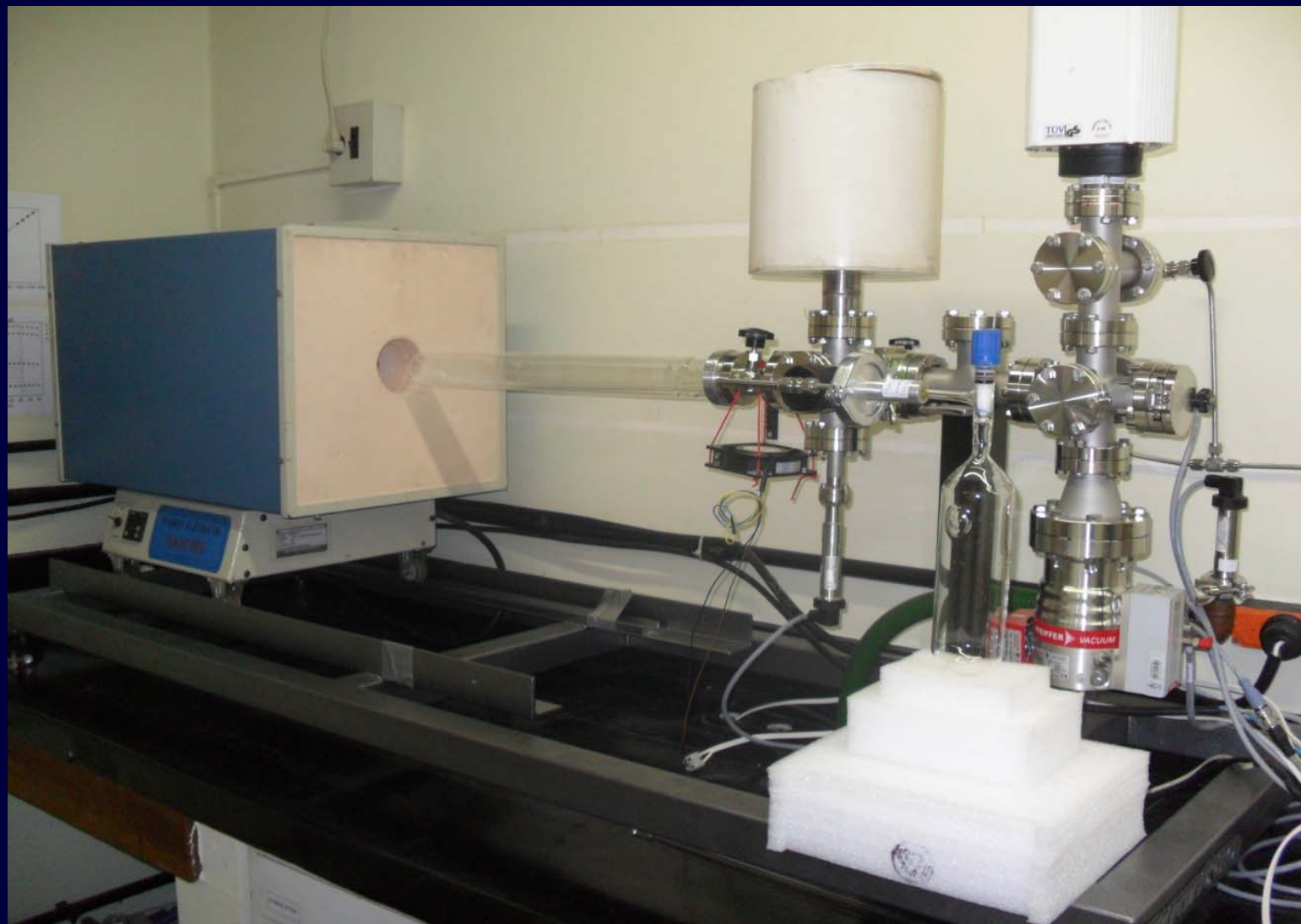
Natural oxygen: 99,759% ^{16}O 0,204% ^{18}O 0,037% ^{17}O

Natural hydrogen: 99,985% ^1H 0,015% $^2\text{H} = \text{D}$



$^{18}\text{O}_2$ 97% ~US\$ 1,000/L D_2^{18}O ~US\$ 2,000/mL

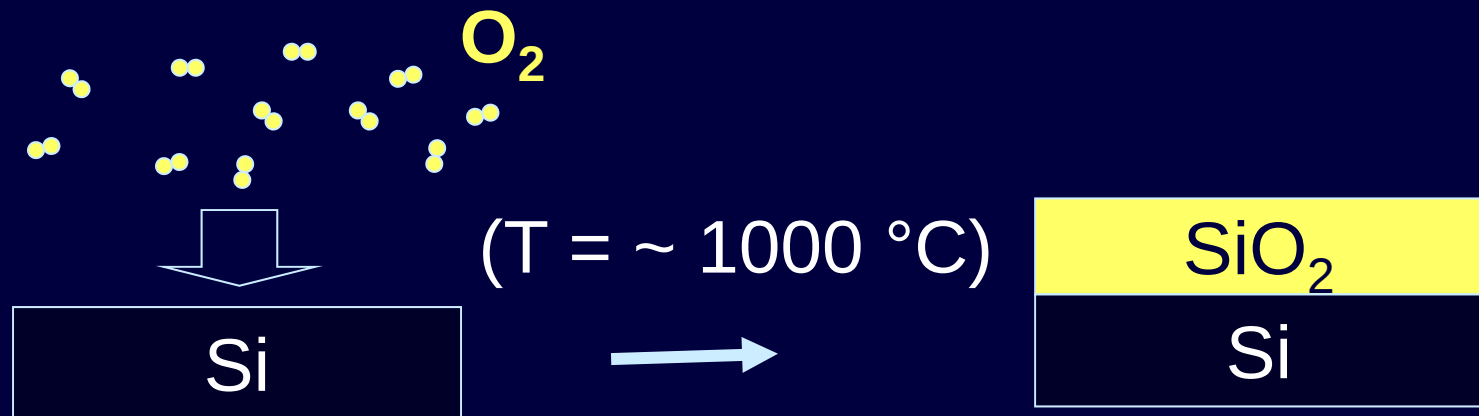
Static atmosphere reactor



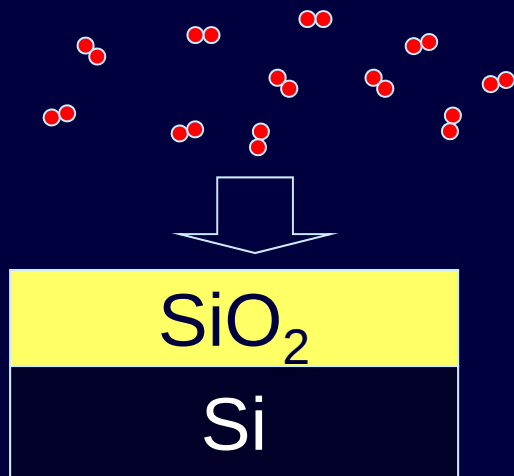
Isotopic Tracing

*Using **isotopes** and **nuclear reactions** to understand the **atomic transport** during the **Si thermal oxidation***

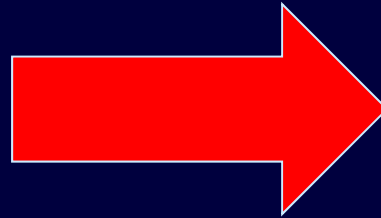
Oxidation in O_2 (natural abundance):



Oxidation in $^{18}\text{O}_2$



$^{18}\text{O}_2$

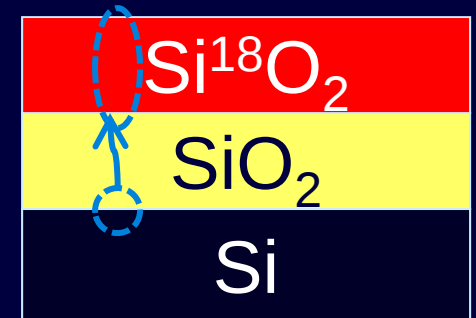
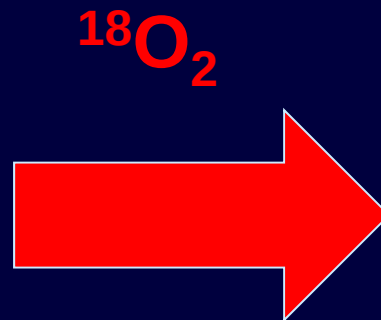
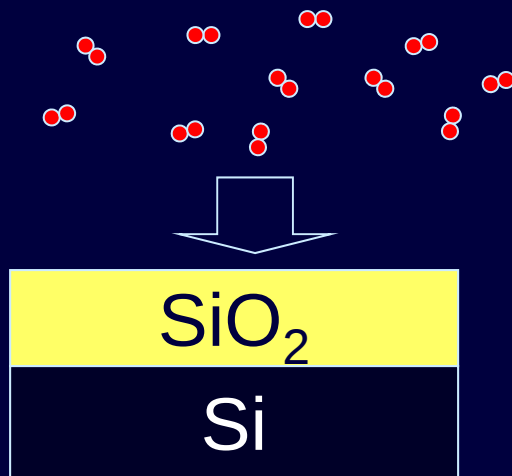


**Who is the mobile species ?
O, Si, or both ?**

Mobile species during oxidation

1st possibility:

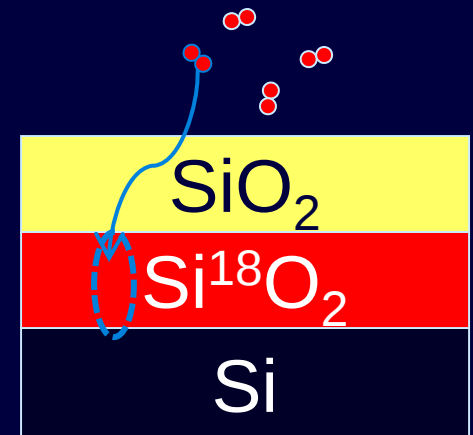
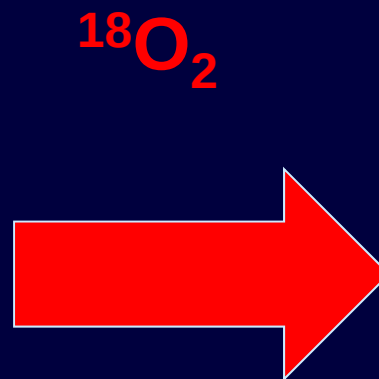
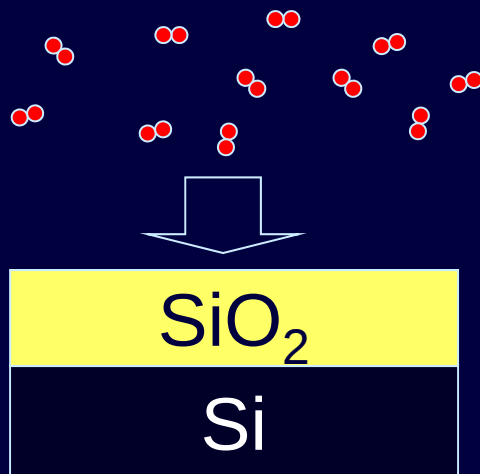
Si is the mobile species



Mobile species during oxidation

2nd possibility:

○ is the mobile species



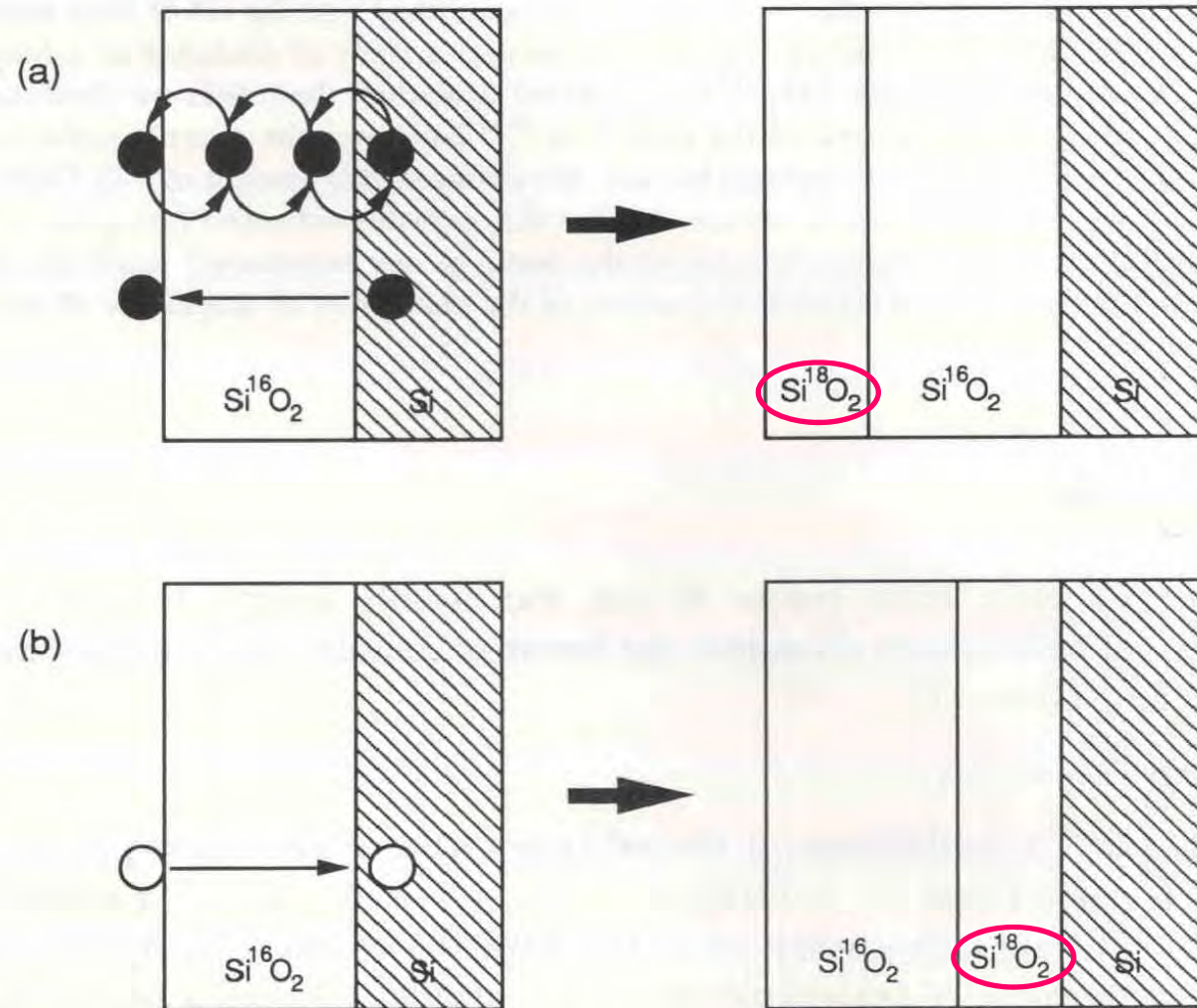
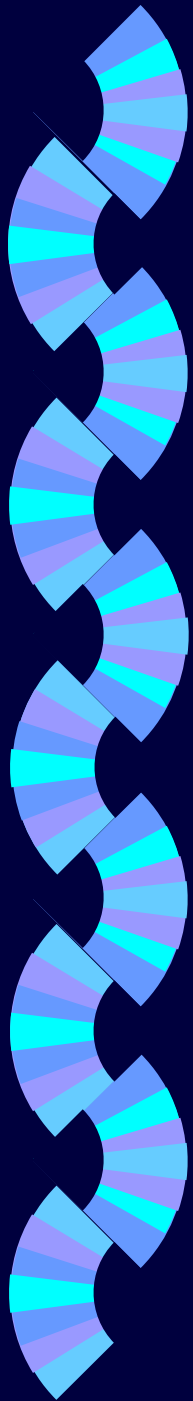


Fig. 1. Illustration of how the labelling technique can be used to identify the mobile species and their transport mechanism: (a) only silicon is mobile; (b) oxygen moves interstitially without reacting with the silica network; (c) oxygen moves interstitially while reacting with the silica network: case of isotopic quasi-equilibrium; (d) oxygen moves interstitially while reacting with the silica network: case far from isotopic quasi-equilibrium; (e) oxygen moves by a simple diffusion of network defects; (f) oxygen moves by migration of network defects in only one direction. Black dots stand for silicon, white dots for oxygen.

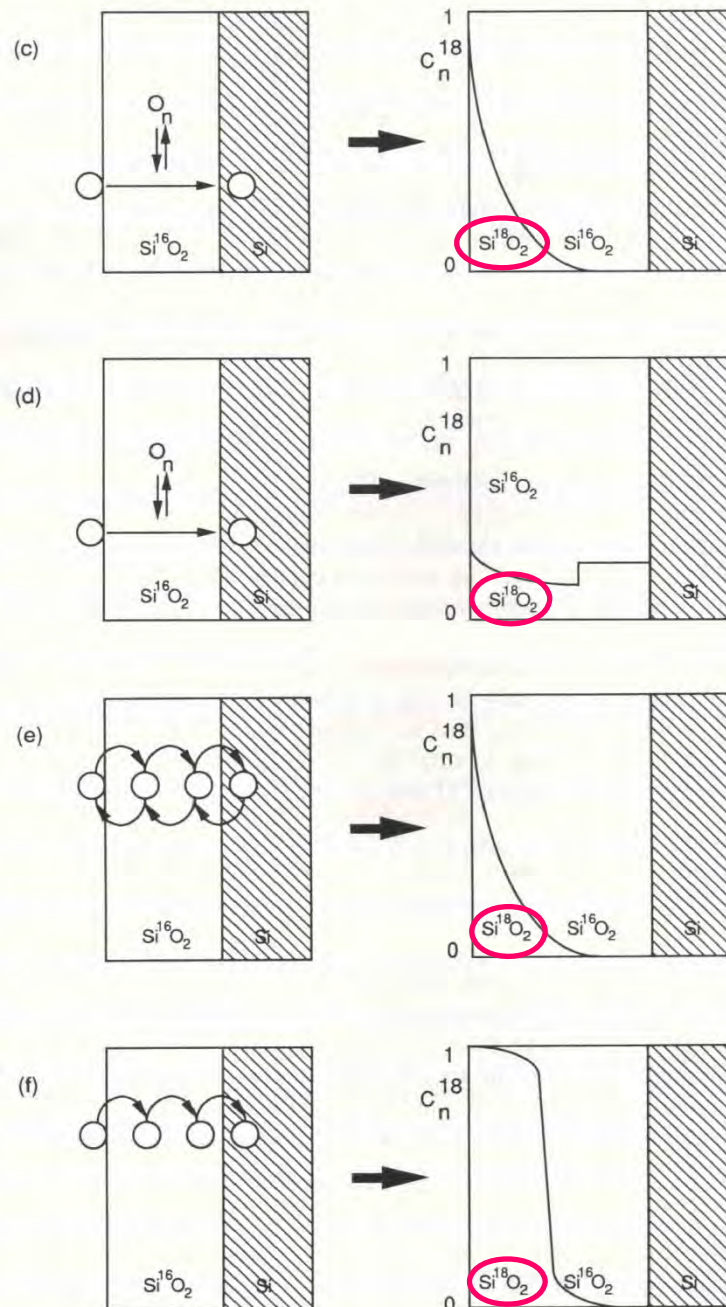
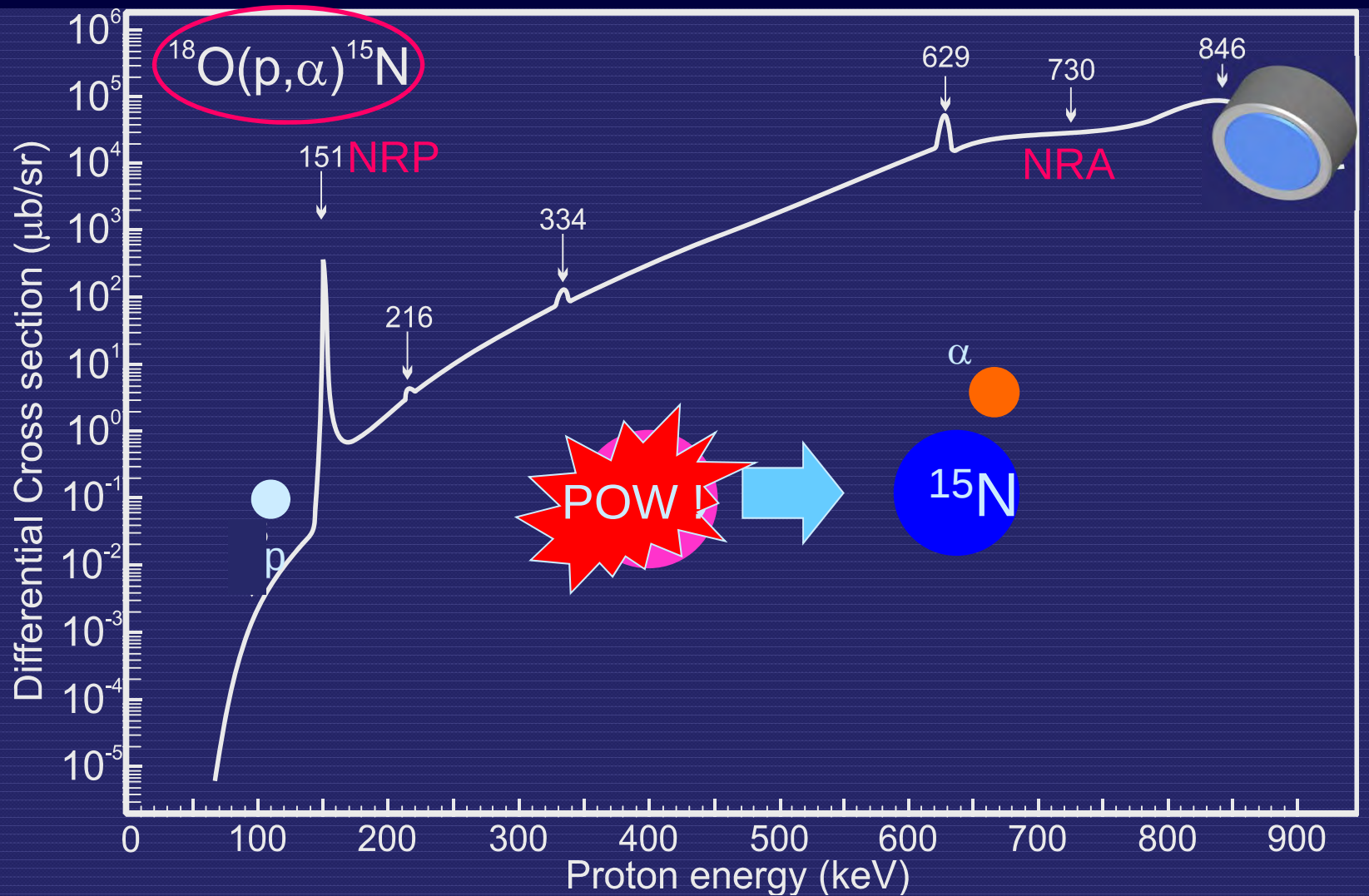
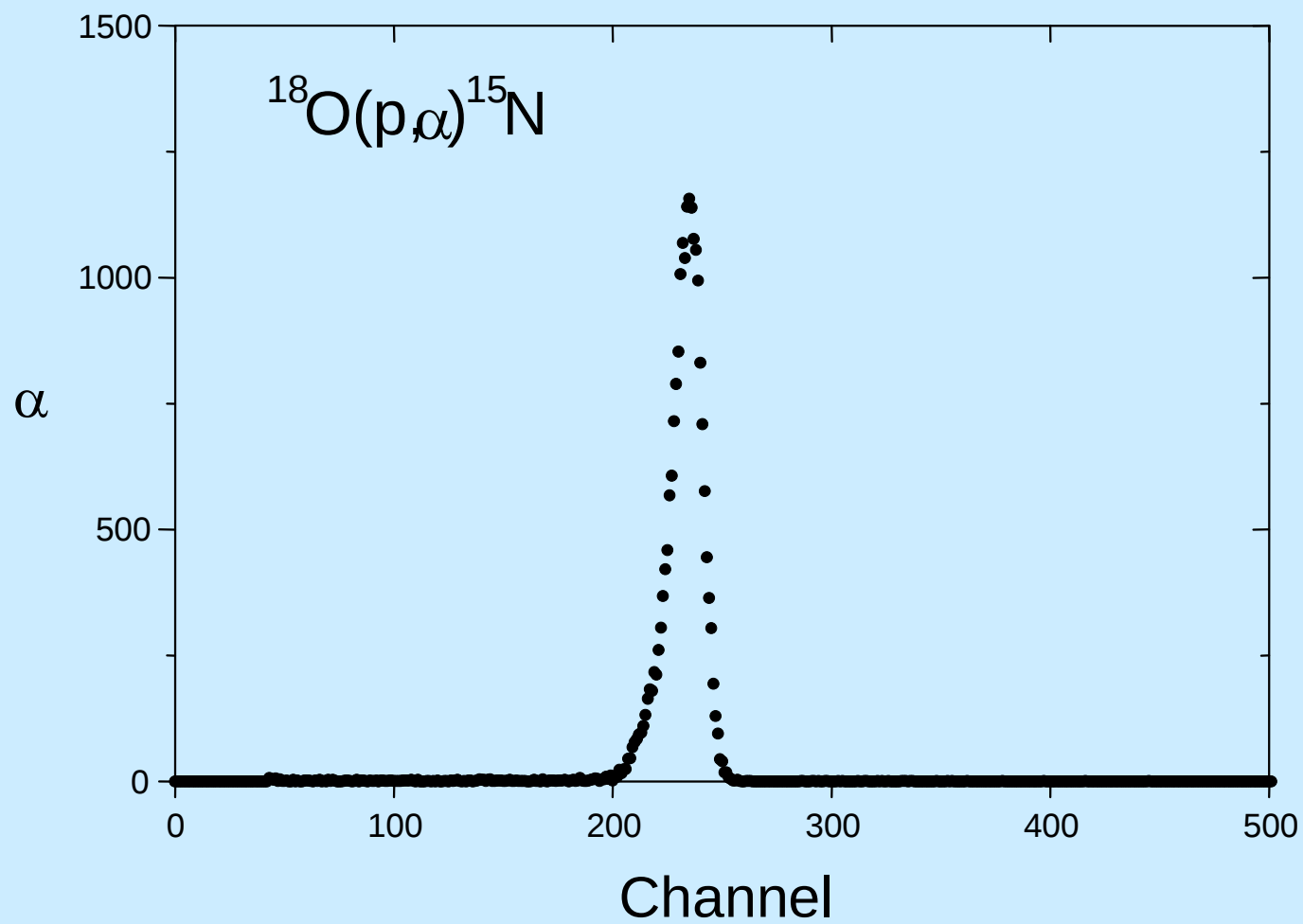
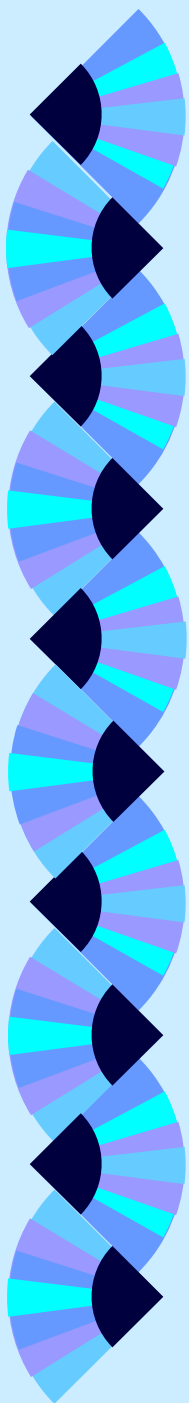


Fig. 1 (continued).

Nuclear Reactions





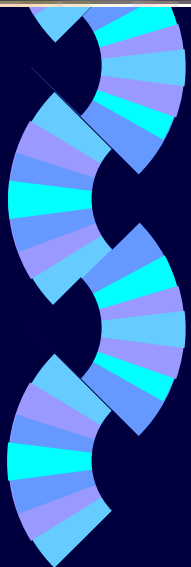


Ion Implantation
Laboratory
UFRGS

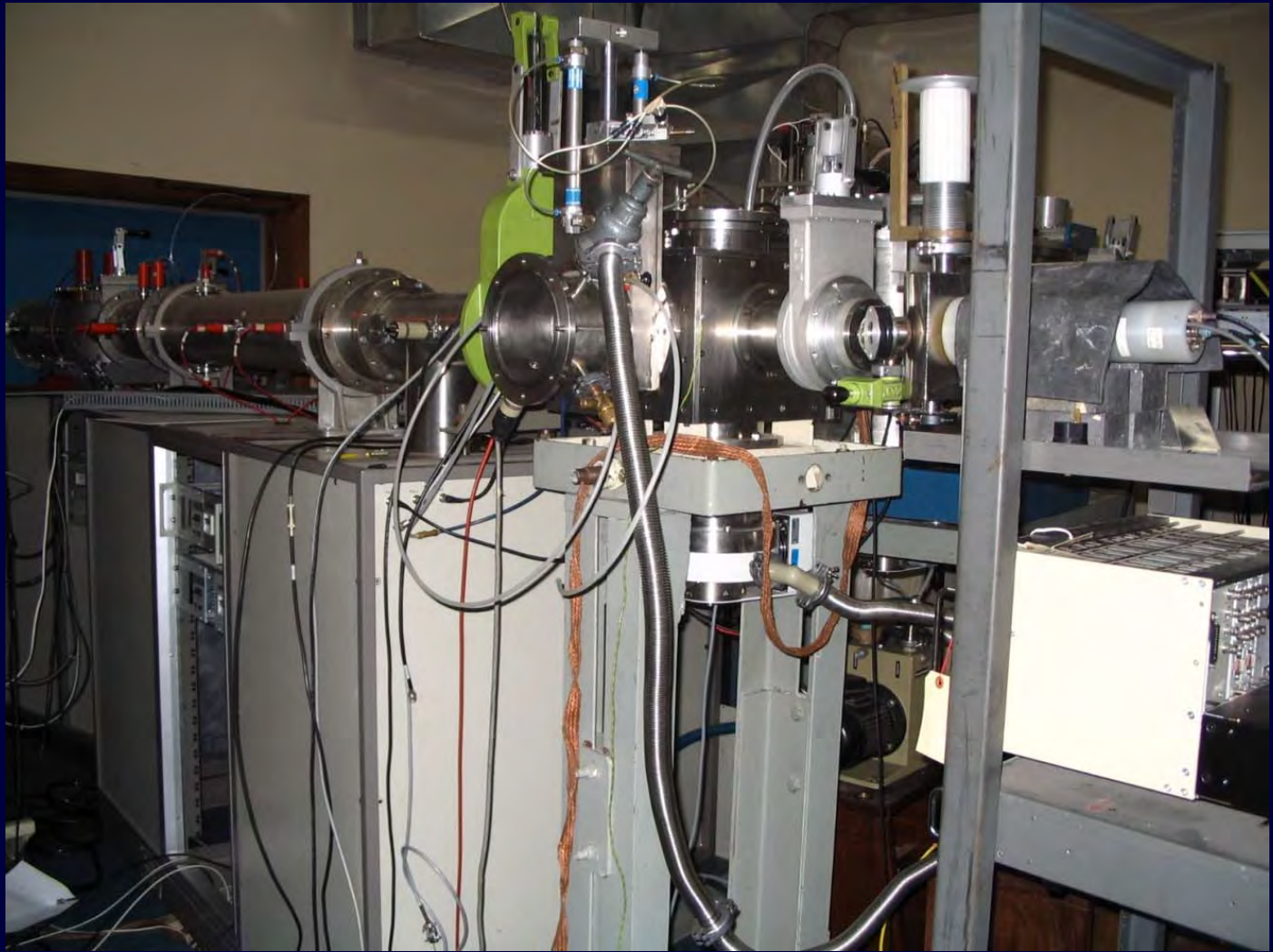
HVEE 3 MV
Tandetron



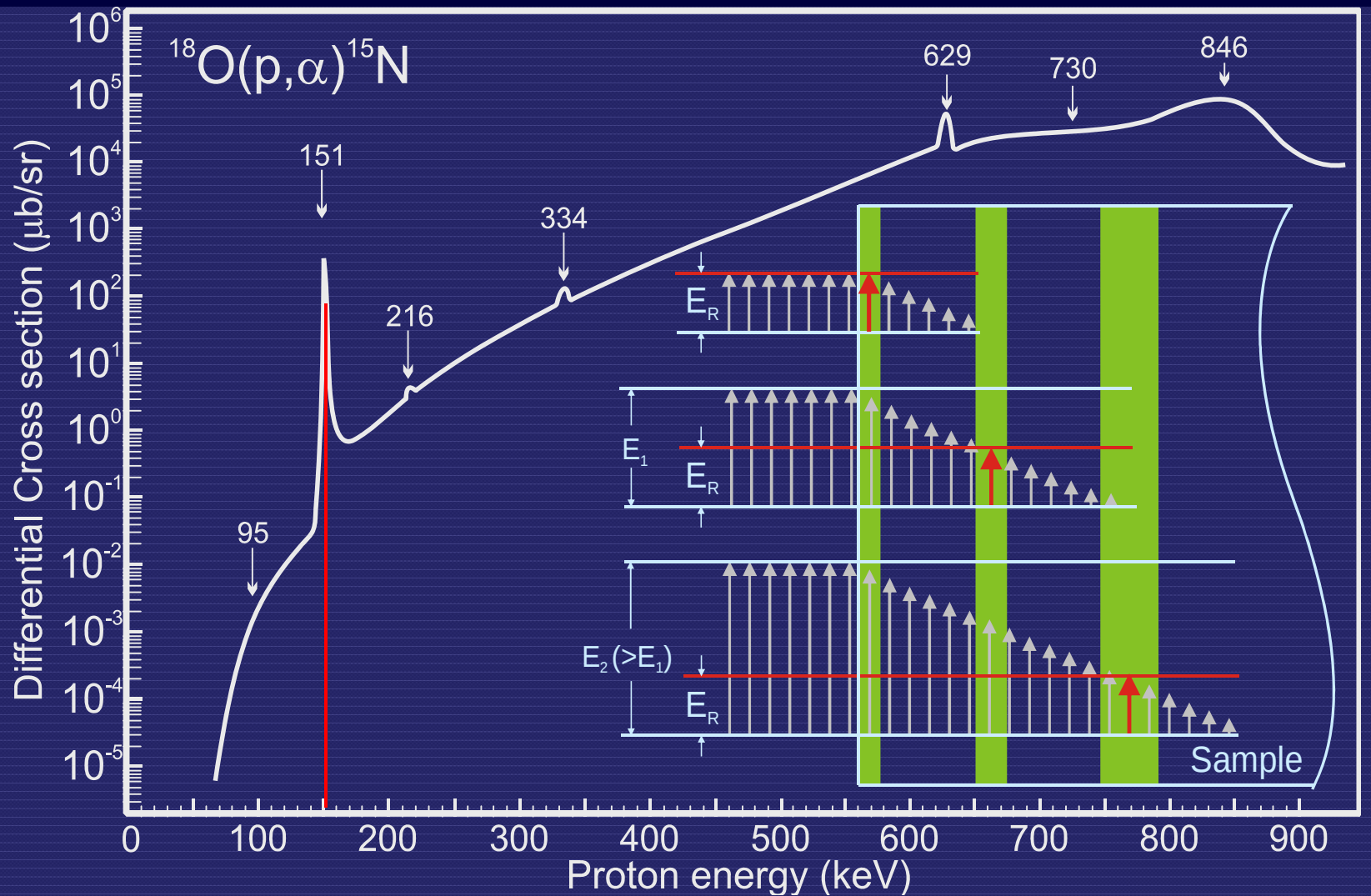
NRA and RBS
chamber

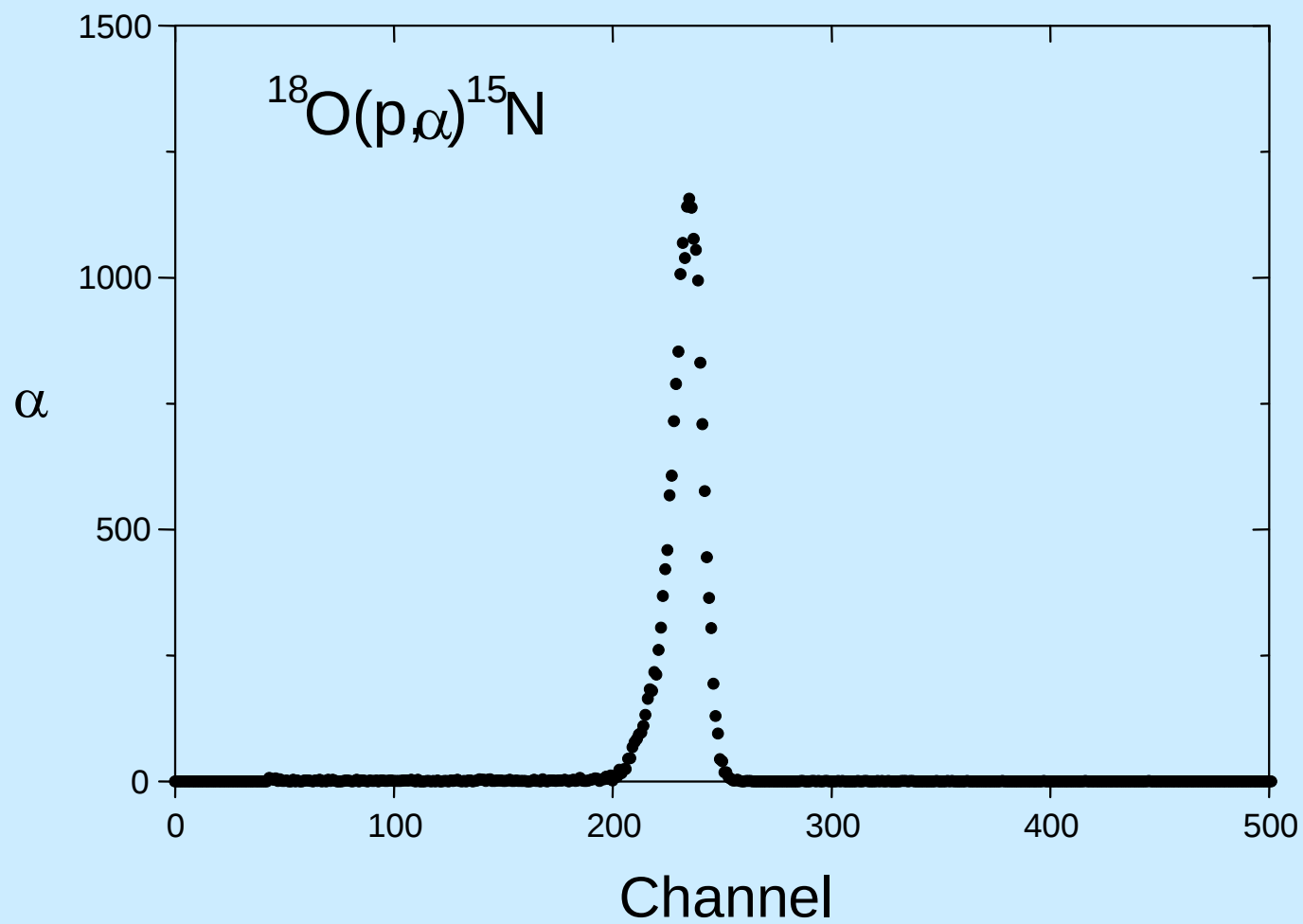
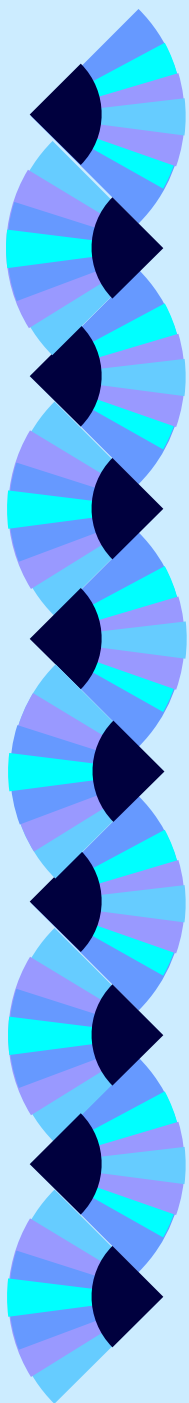


HVEE 500 kV single-ended: NRP chamber

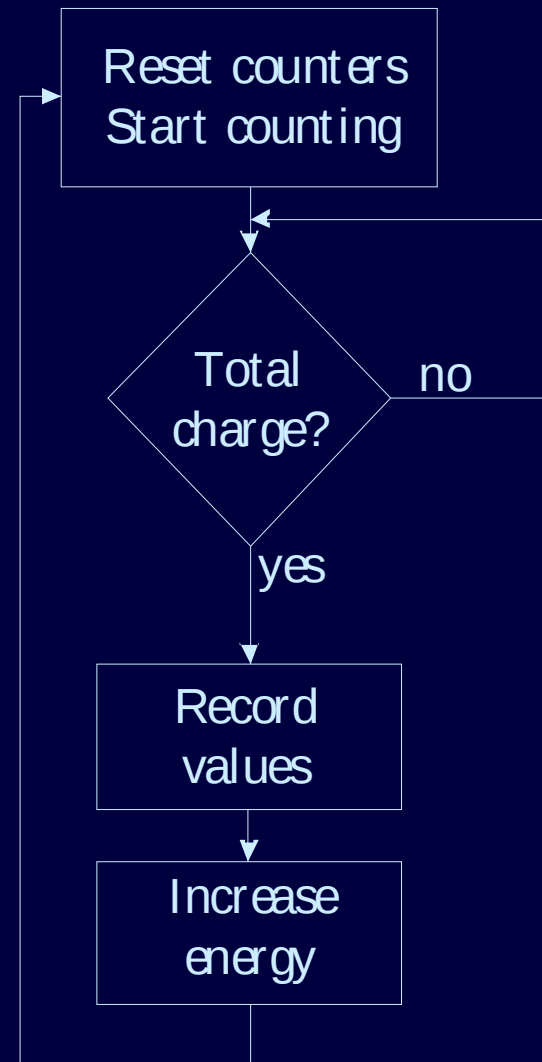
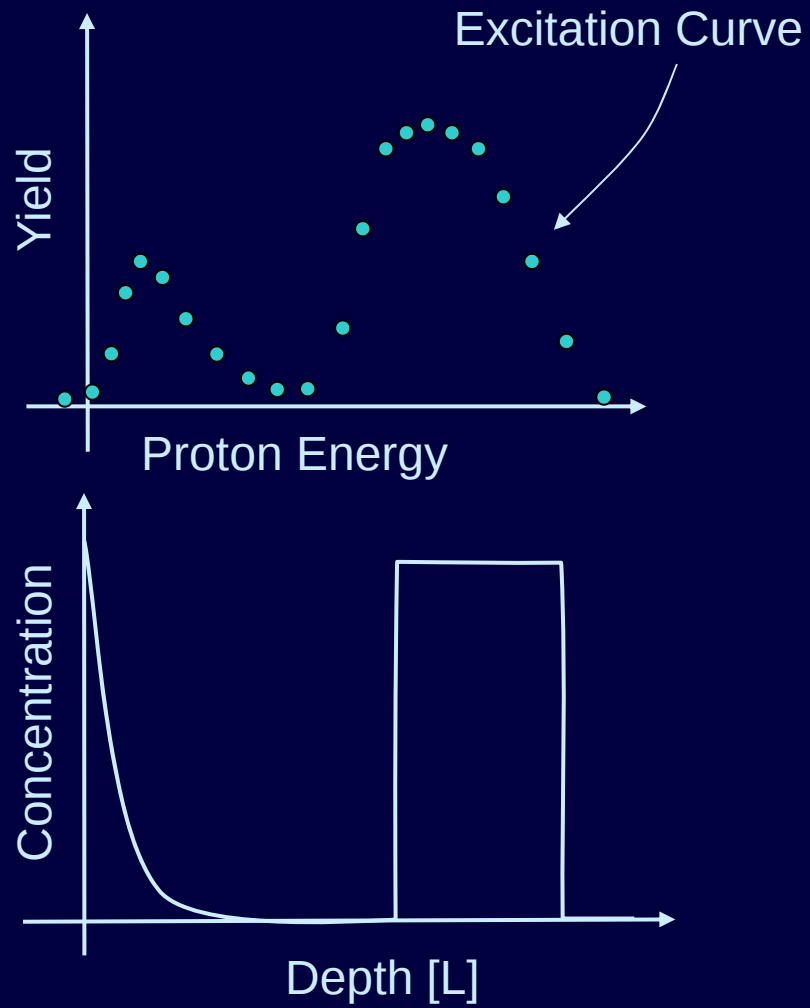
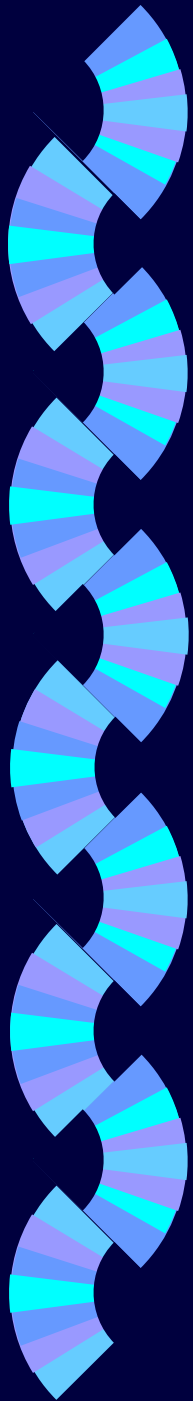


Resonant Nuclear Reaction - NRP

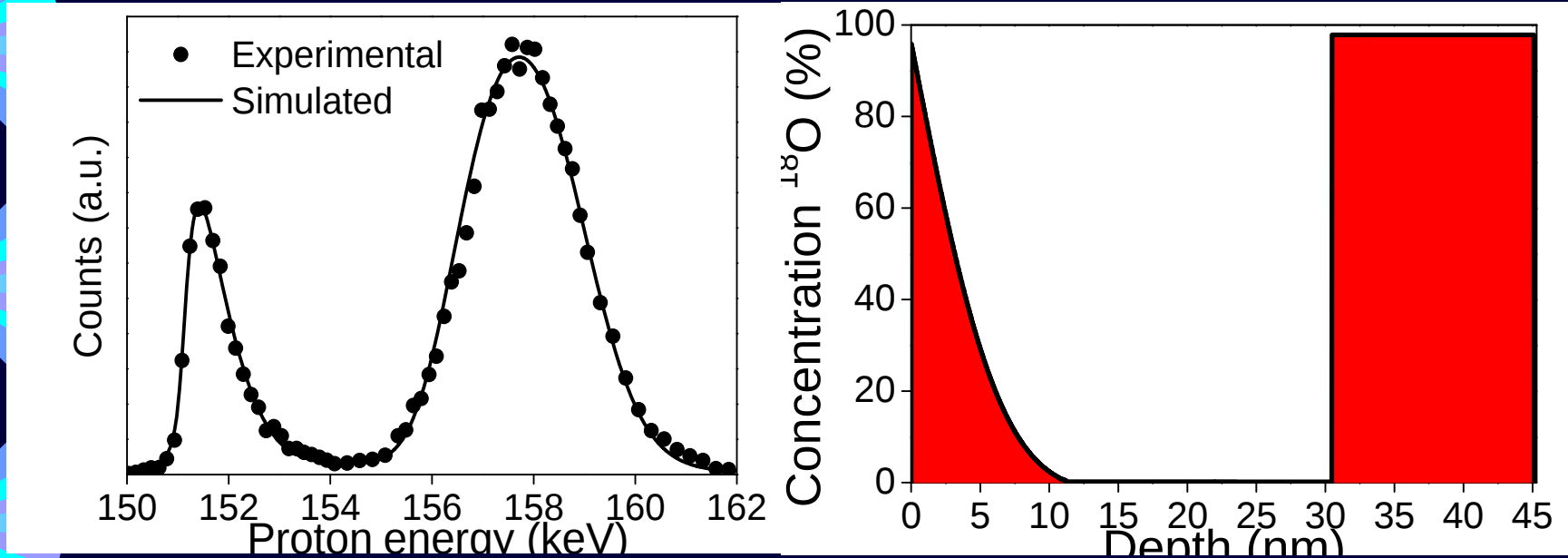




Profile

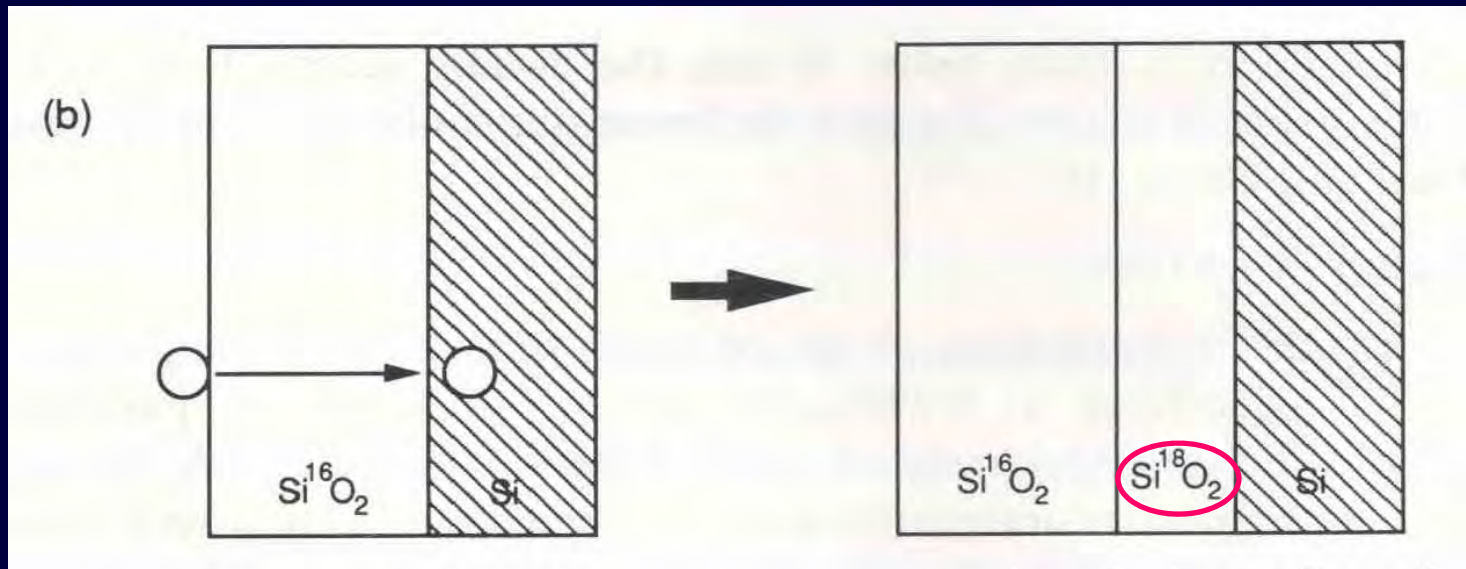
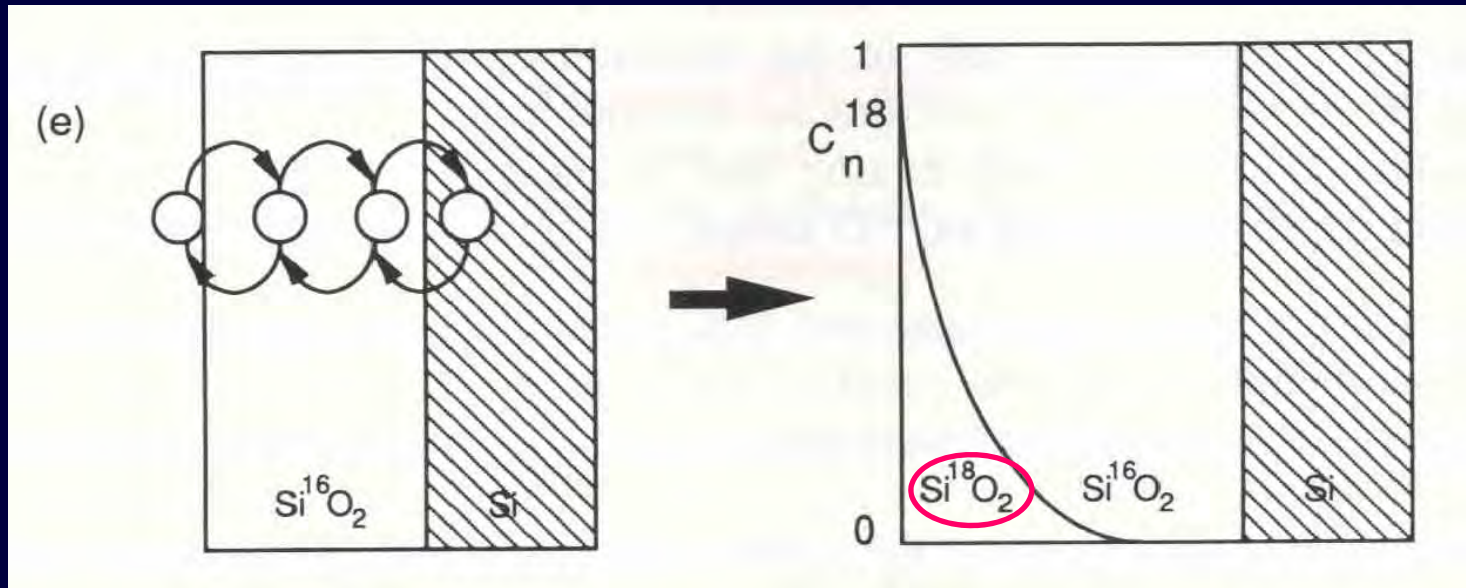
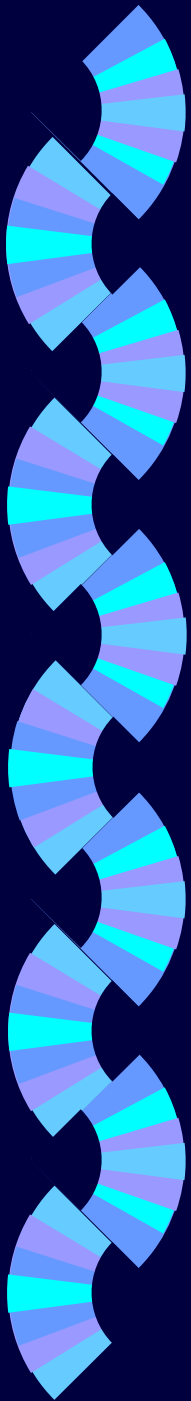


$^{16}\text{O}_2$ - $^{18}\text{O}_2$ @ 1000°C



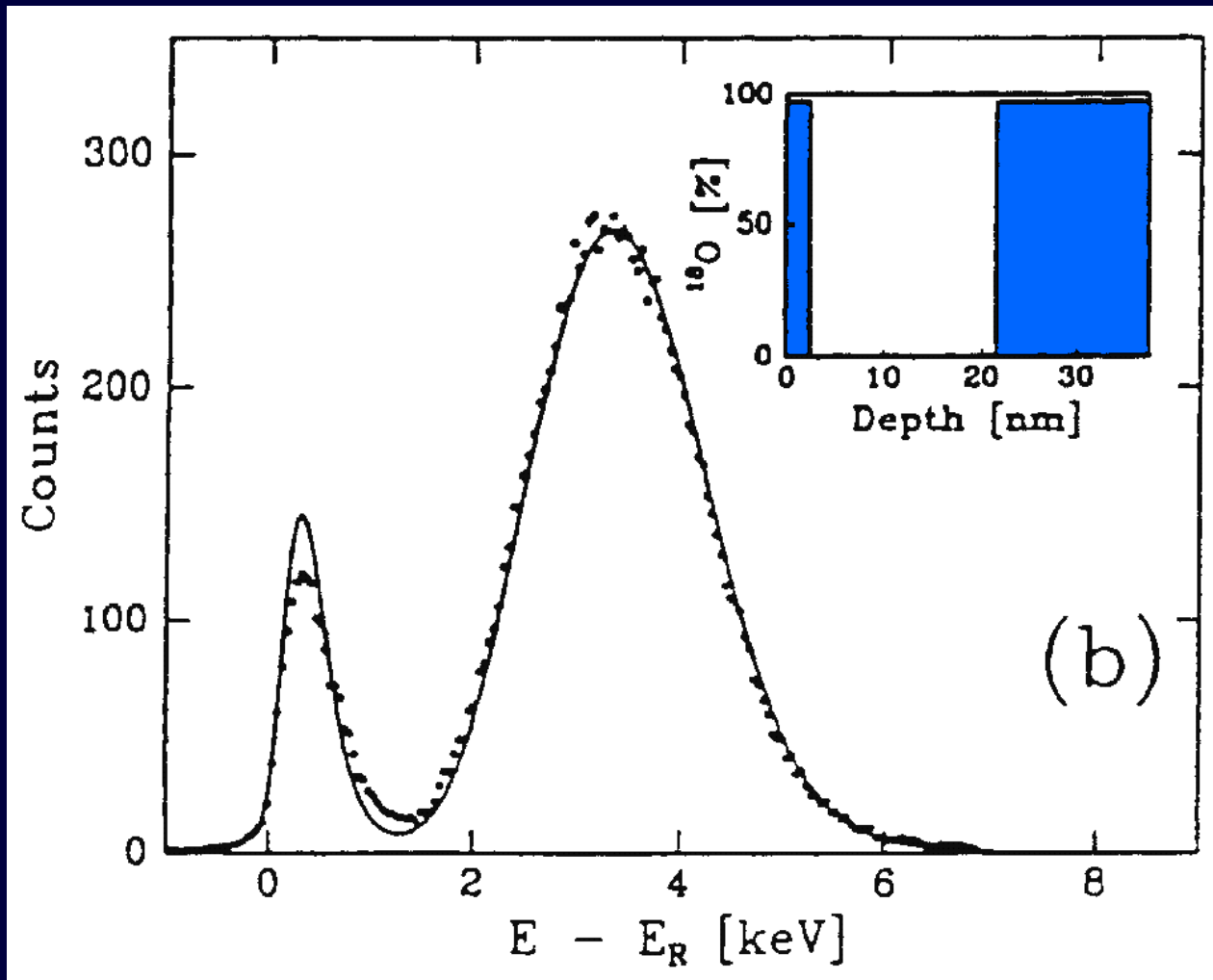
Oxygen is the mobile species and it is transported:

- By diffusion, interacting with network defects (surface region)
- Interstitially, without reacting with the oxide already formed, until reaching the semiconductor interface and there reacting forming SiO_2

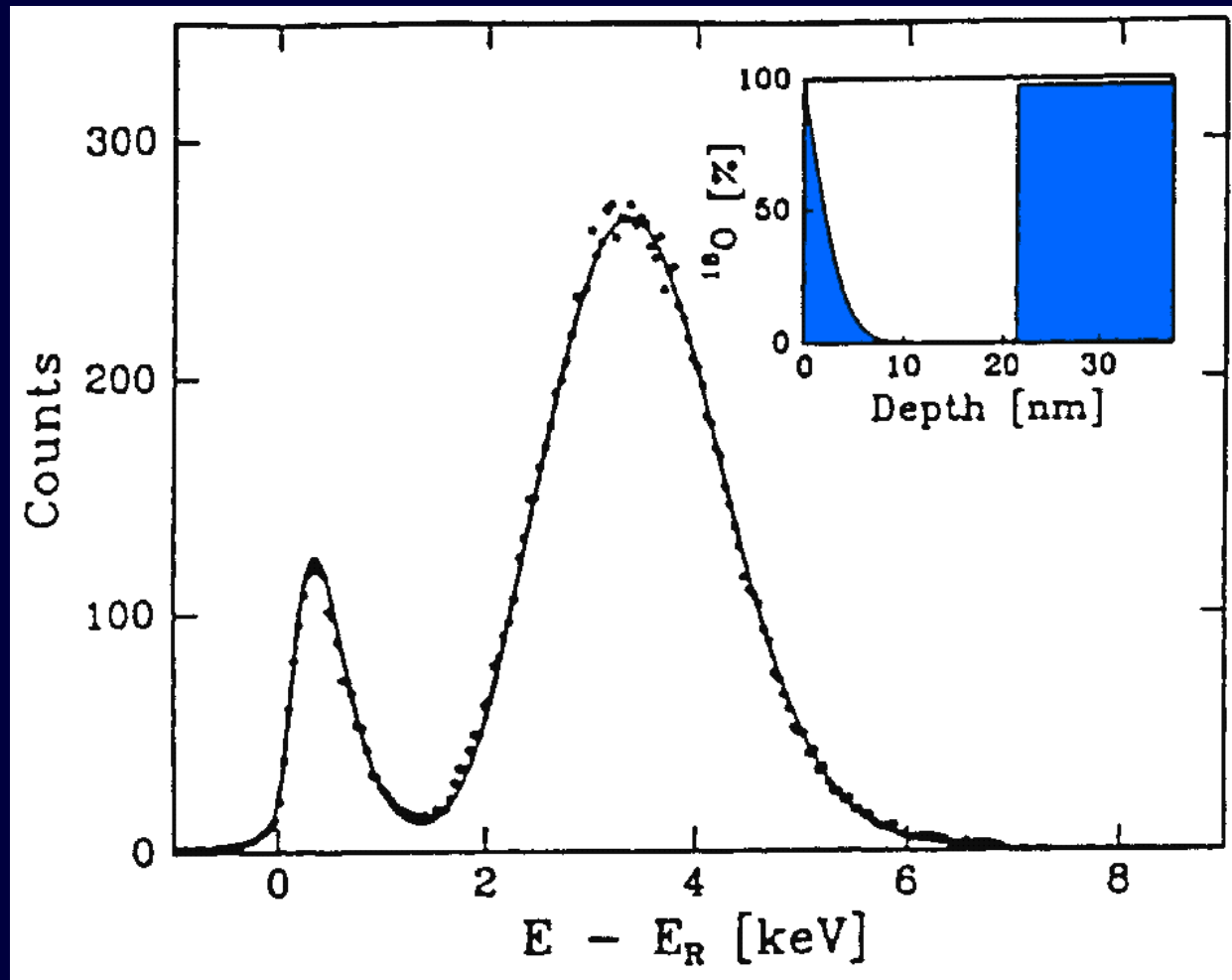


Profiles reliability

$^{16}\text{O}_2$ - $^{18}\text{O}_2$ @ 1000°C



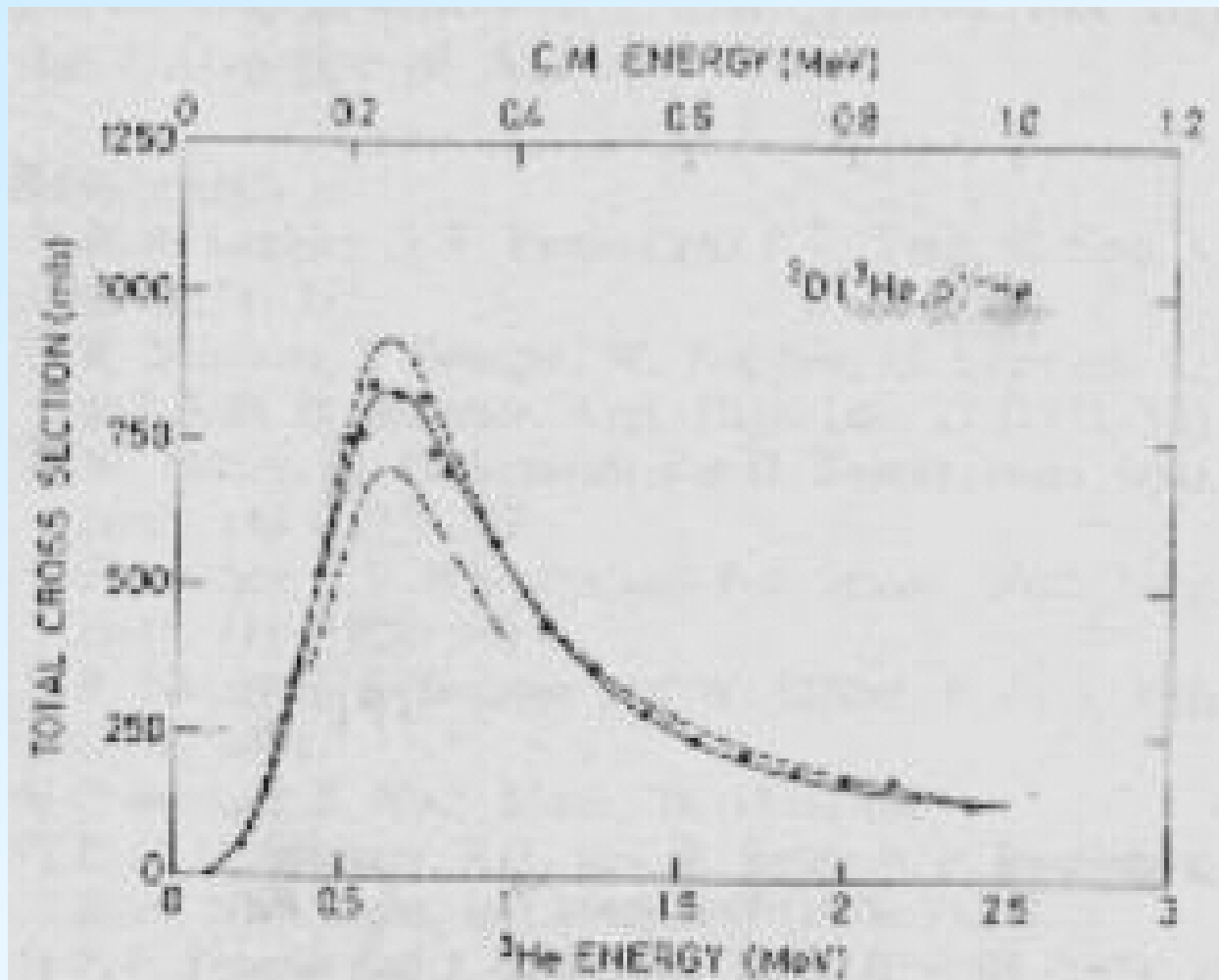
$^{16}\text{O}_2$ - $^{18}\text{O}_2$ @ 1000°C



Trimaille et al. The Physics and Chemistry of SiO_2 and the Si- SiO_2 Interface - 3, ed. H.Z. Massoud *et al.* (The Electrochemical Society, Pennington, 1996), vol.96-1, p. 59

NRA: D amounts

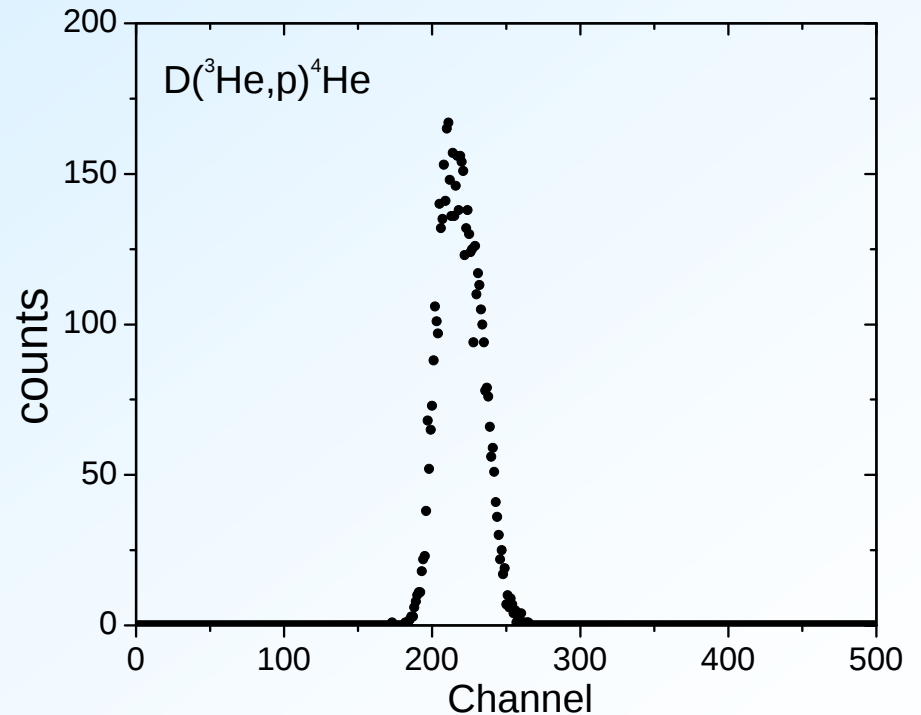
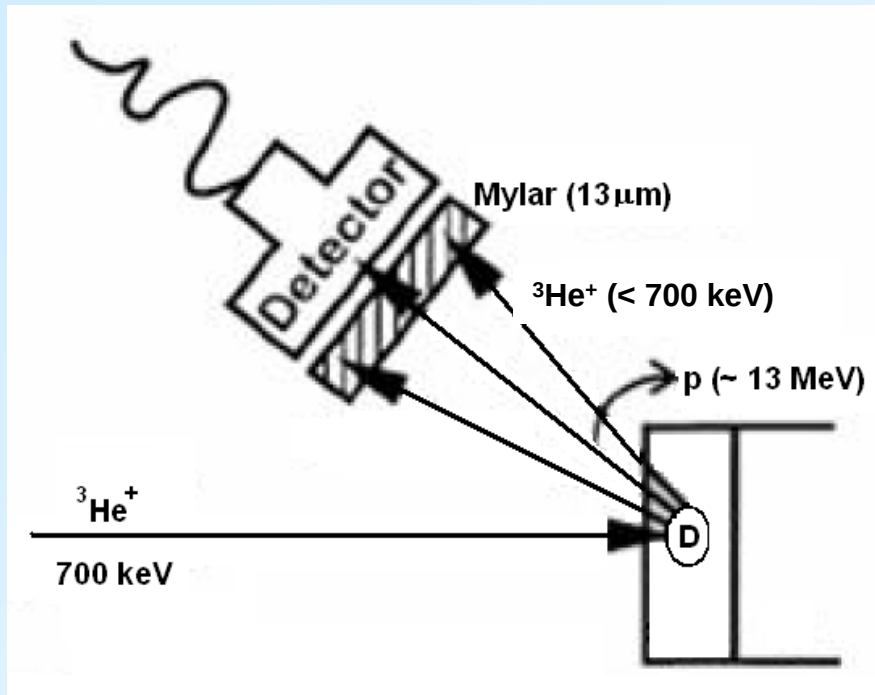
- Nuclear reaction $D(^3\text{He},p)^4\text{He}$: plateau ~ 700 keV
- Total amounts of D (insensitive to H)



D. Dieumegard *et al.*
NIM 166 (1979) 431

D quantification

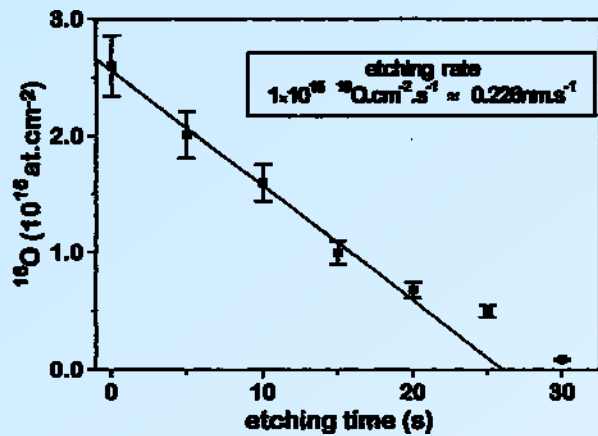
$D(^3\text{He},p)^4\text{He}$ at 700 keV



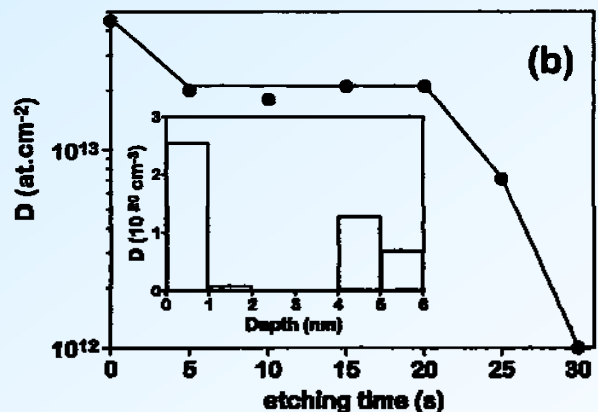
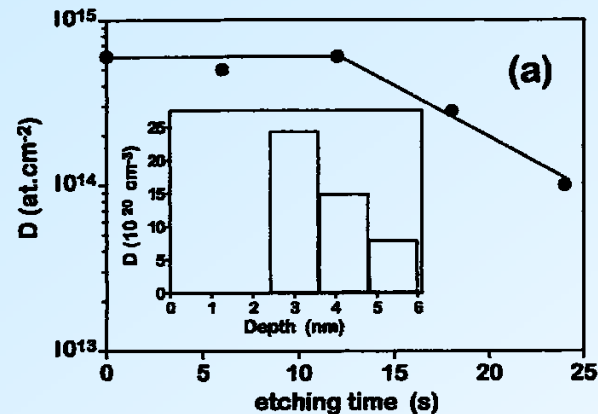
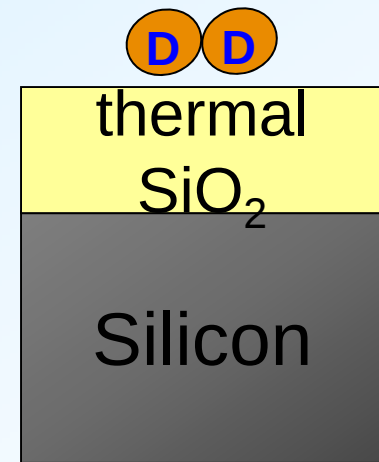
sensitivity $\sim 4.0 \times 10^{12} \text{ D.cm}^{-2}$

Accuracy: 10%

D profiling



Areal density of ^{16}O in oxide films versus time of chemical dissolution in dilute HF-solution

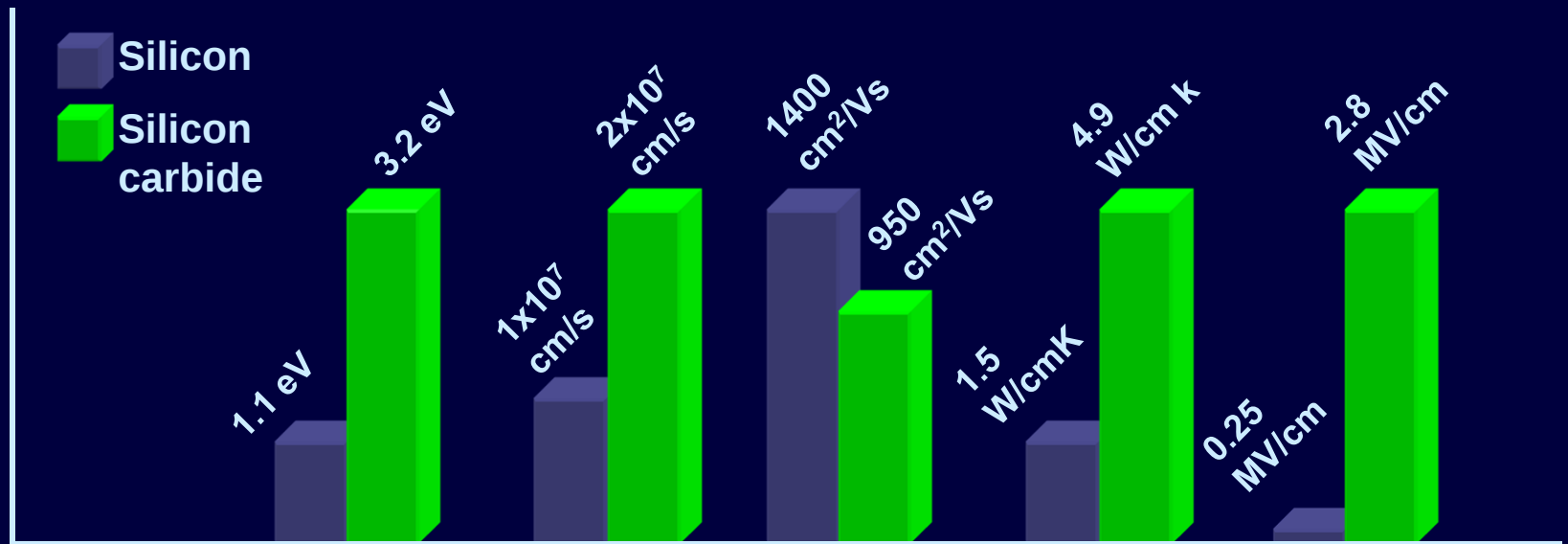


Areal densities of deuterium in oxide films versus time of chemical etching. The corresponding **D profiles** determined by differentiation are shown in the insets: **(a)** as-loaded with D_2 ; **(b)** loaded with D_2 and annealed in vacuum at 650°C for 30 min

I.J.R. Baumvol, **F.C. Stedile**, C. Radtke, F.L. Freire, Jr., E.P. Gusev, M.L. Green, D. Brasen, Nucl. Instrum. Meth. B 204

I.J.R. Baumvol, E.P. Gusev, **F.C. Stedile**, F.L. Freire, Jr., M.L. Green, D. Brasen, Appl. Phys. Lett. 72 (1998) 450.

Motivation SiC



Bandgap

Saturation
drift velocity

Mobility

Thermal
conductivity

Breakdown
el. field*

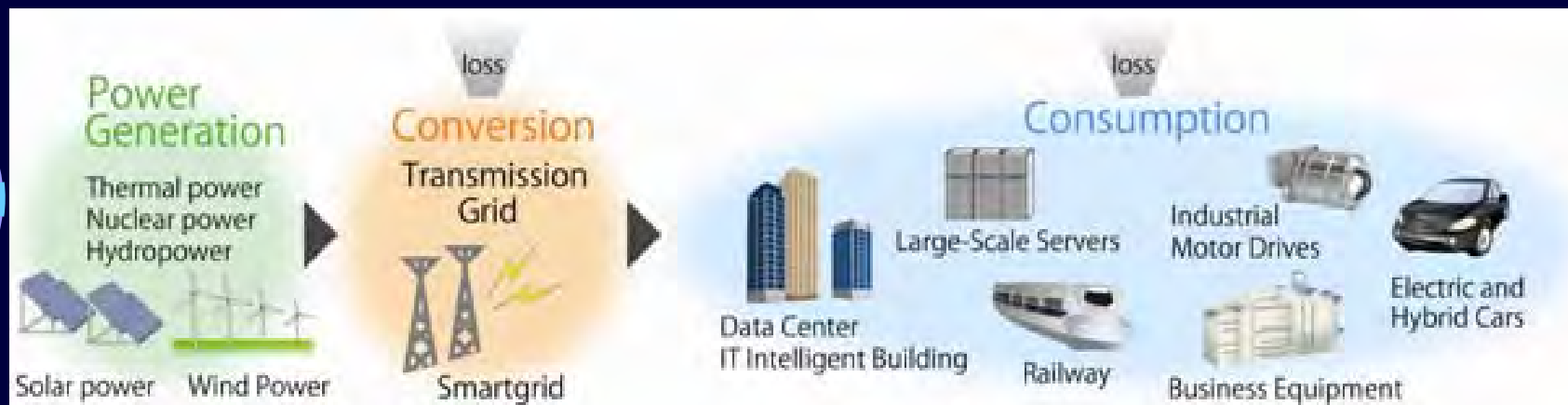
High temperature
Electronics and sensors

* for 1200 V blocking voltage

High frequency
power devices

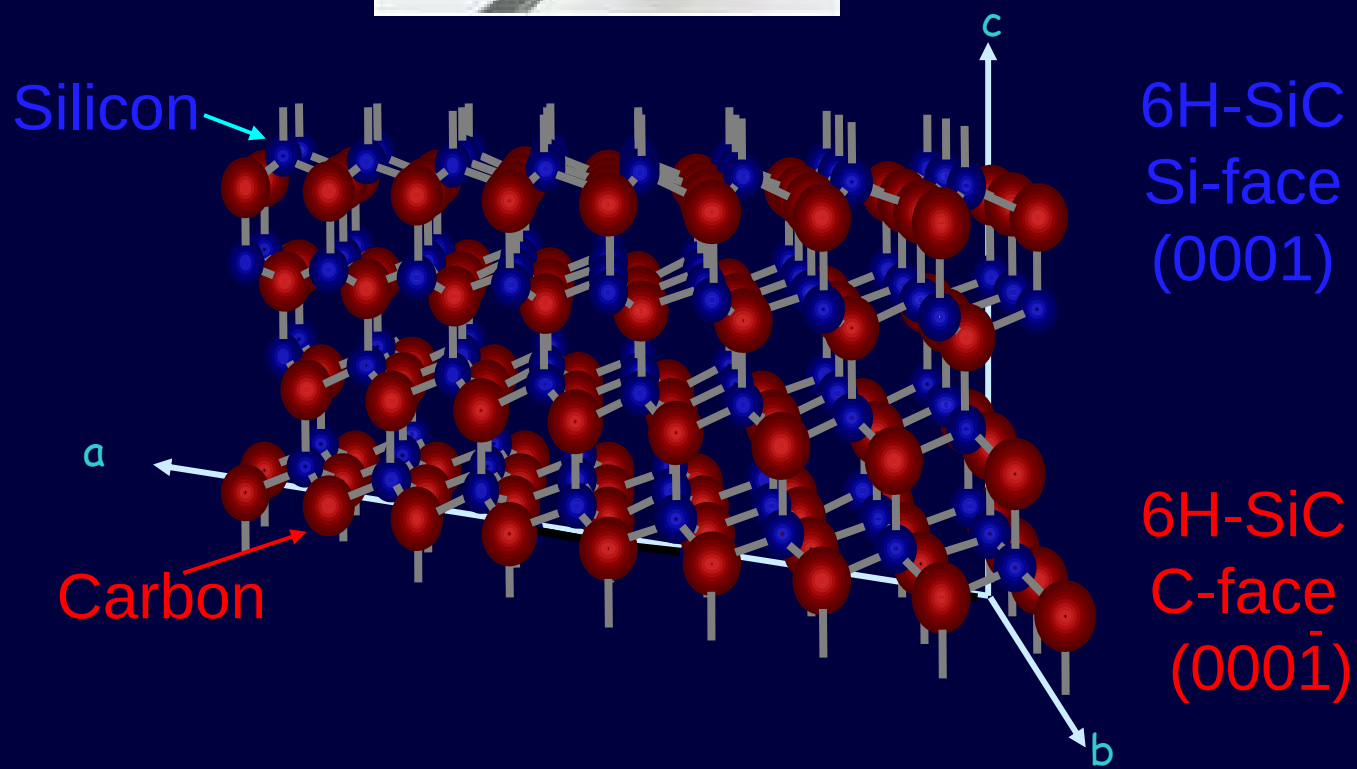
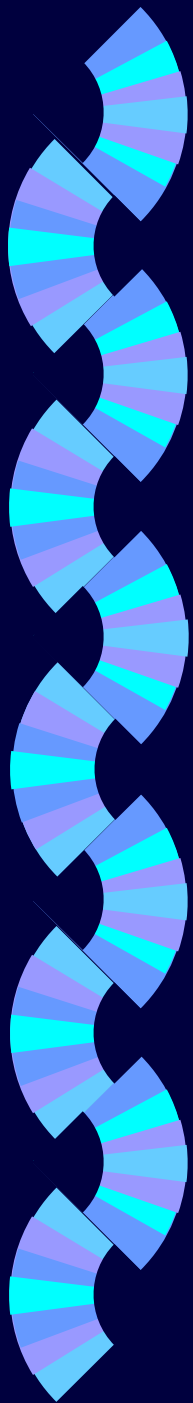
Low and medium voltage
high switching frequency
power electronic devices with
low losses

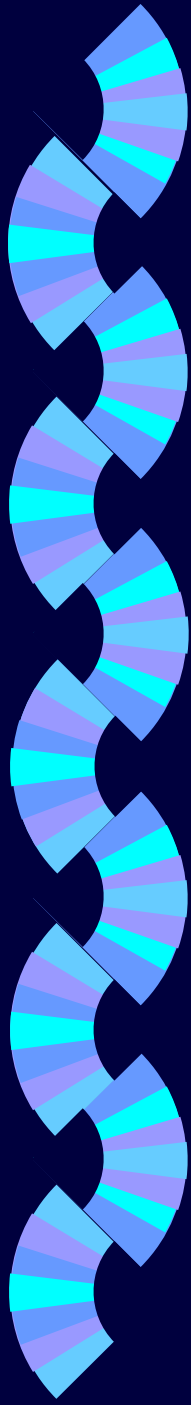
SiC < energy loss than Si



Japan: 1% less energy loss
Si bipolar transistor → SiC MOSFET
85% less energy loss

Energy save ~ 4 nuclear power plants

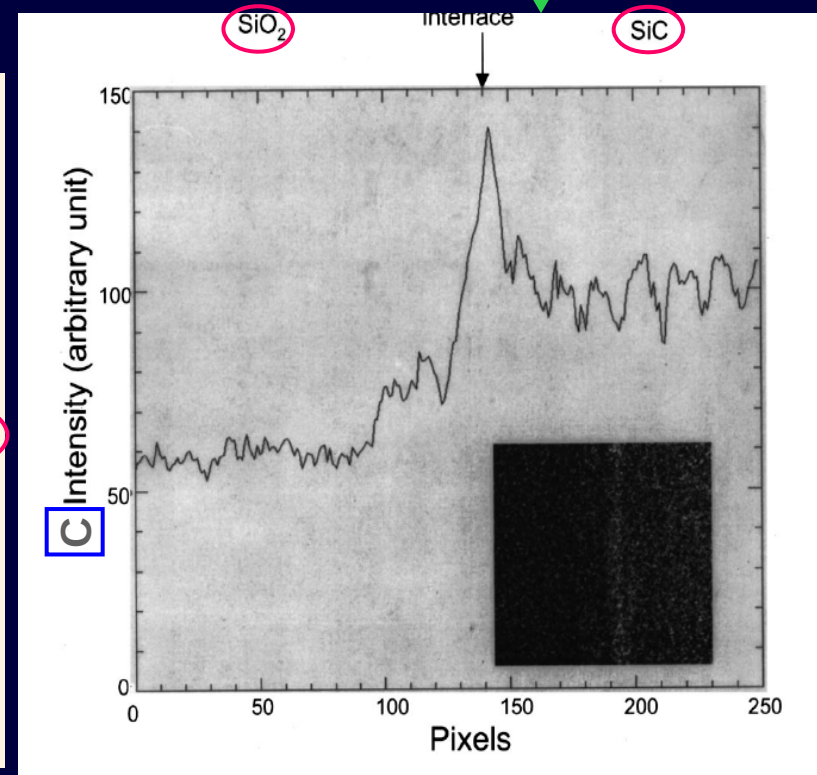
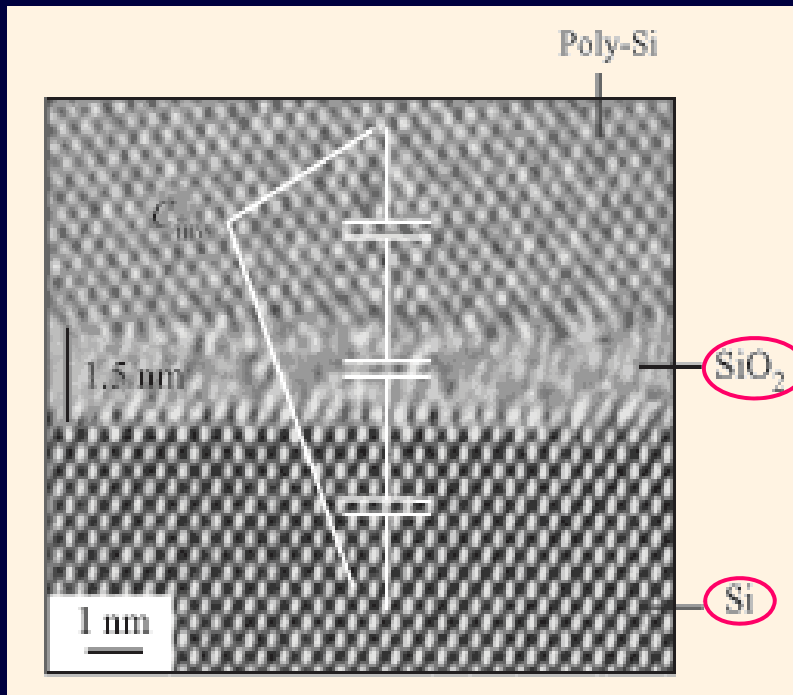




SiC : only compound semiconductor on which it is possible to thermally grow a SiO_2 film, as on Si

➤ Part of SiO_2/Si technology can be transferred or adapted to the SiO_2/SiC technology

Interfaces: SiO_2/Si x SiO_2/SiC



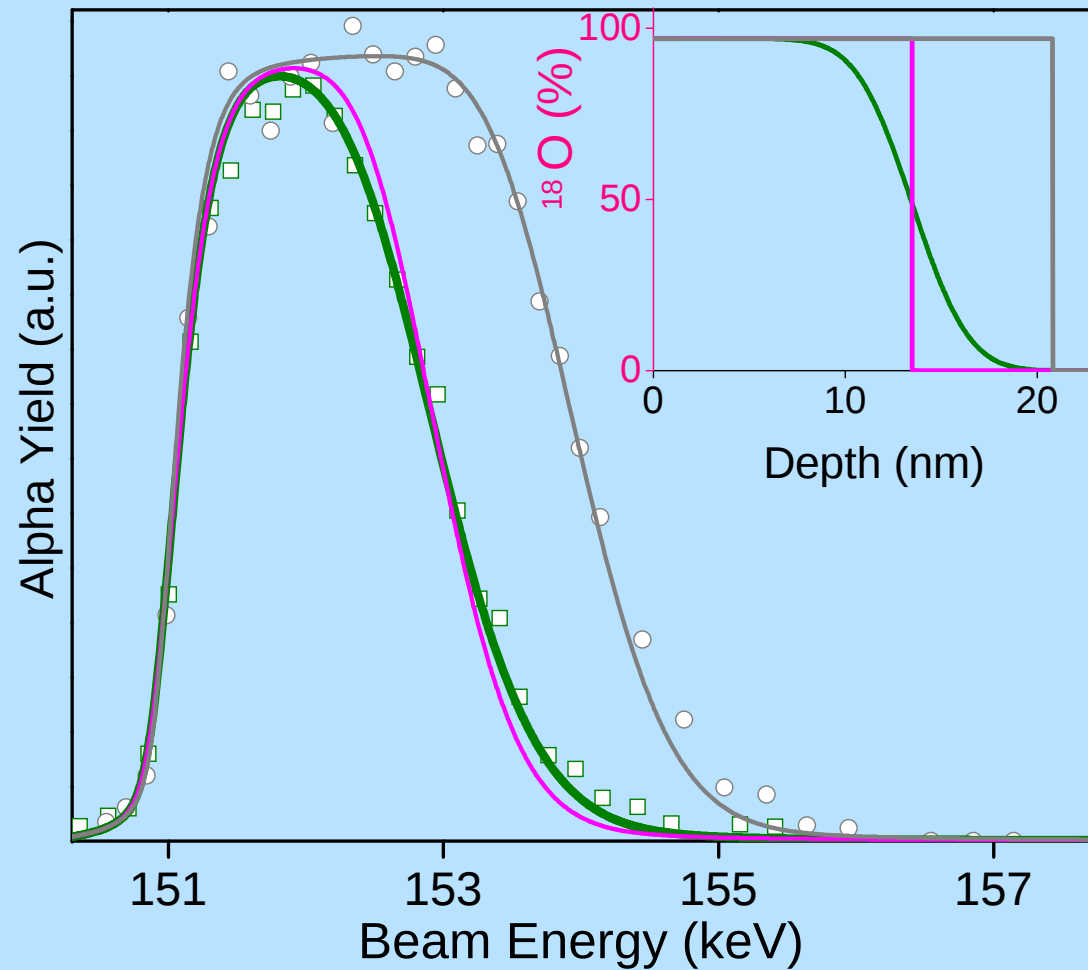
<http://www.ibm.com>

K.C. Chang *et al.* Appl. Phys. Lett. 77 (2000) 2186

Single step in $^{18}\text{O}_2$

SiO_2 / SiC / Si

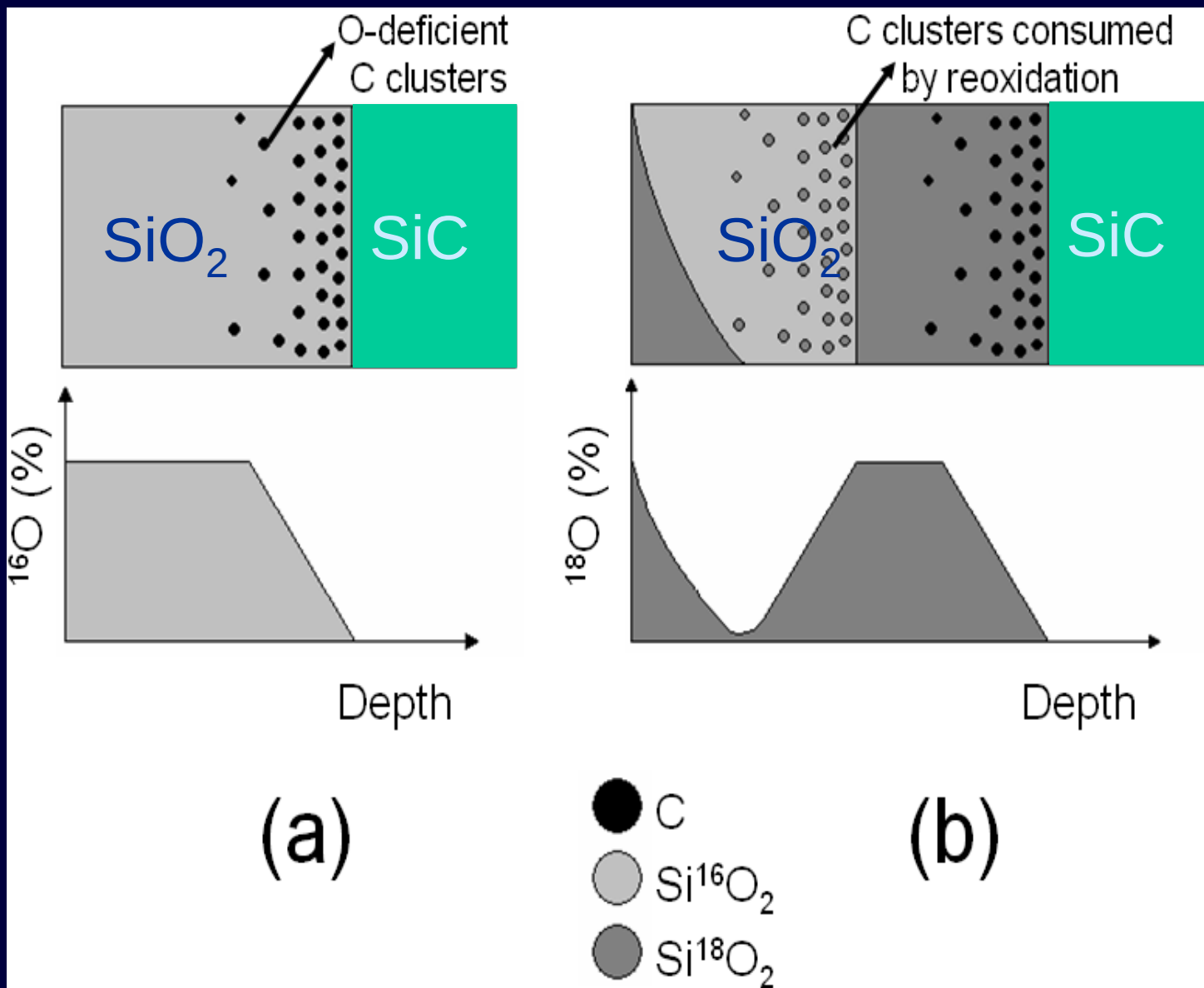
○ Si
□ SiC
Si-face



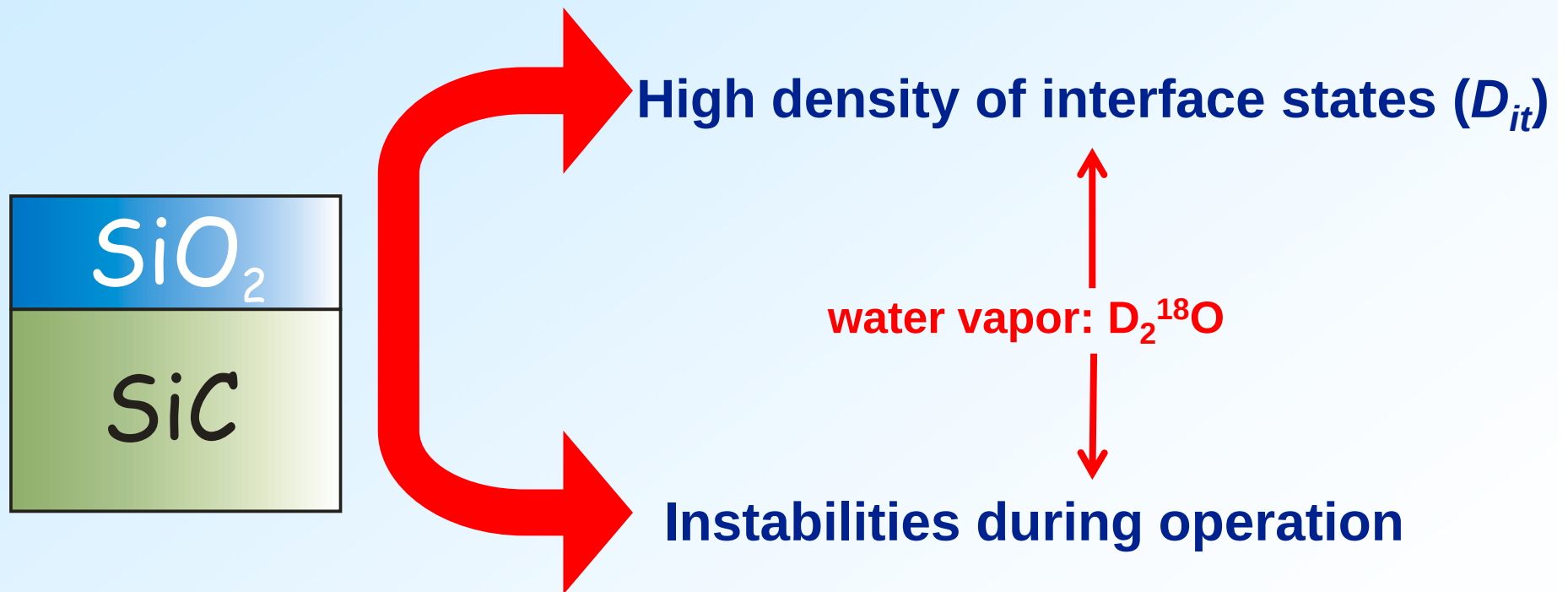
C. Radtke, I.J.R. Baumvol, B.C.Ferrera, **F.C. Stedile**
Appl. Phys.Lett. 85 (2004) 3402

oxidation in $^{16}\text{O}_2$

oxidation in $^{16}\text{O}_2 + ^{18}\text{O}_2$



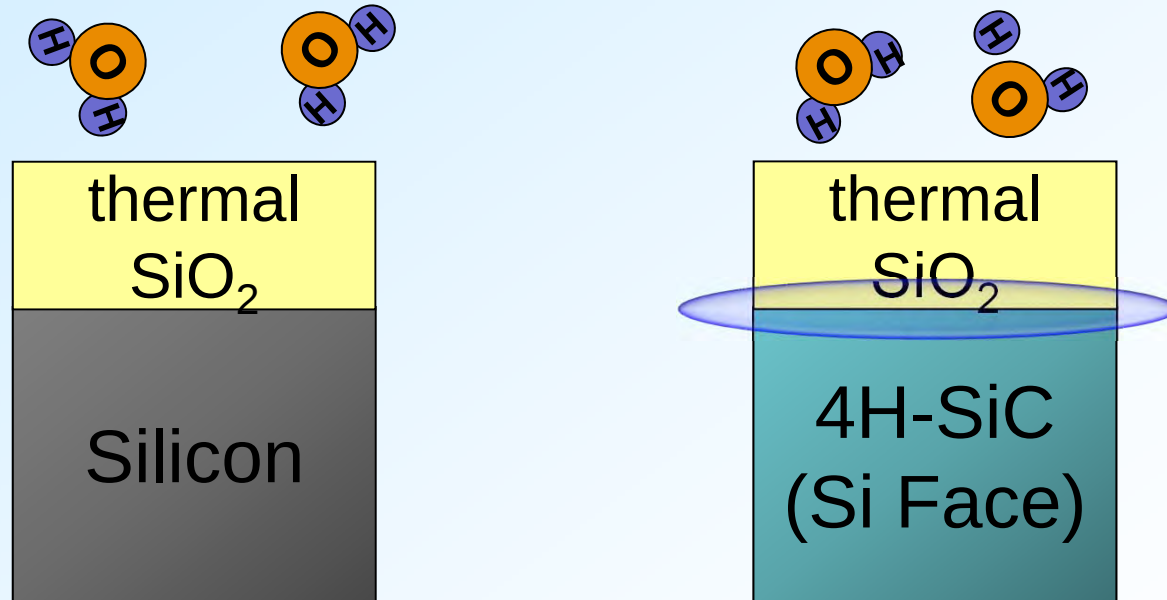
Challenges in SiC technology



Water related problems in SiO_2/Si

- ❑ humidity in a clean room fabrication facility is between 30 and 50%
- ❑ negative oxide charge buildup near the SiO_2/Si interface
- ❑ increases in the interface state density already reported for SiO_2/Si
- ❑ negative-bias-temperature instabilities attributed to water related species at the SiO_2/Si interface

Water vapor incorporation in SiO_2/SiC and SiO_2/Si



G.V. Soares, I.J.R. Baumvol, S.A. Corrêa, C. Radtke, **F.C. Stedile**
Appl. Phys.Lett. 95 (2009) 191912

G.V. Soares, I.J.R. Baumvol, S.A. Corrêa, C. Radtke, **F.C. Stedile**
Electrochemical and Solid-State Letters 13 (2010) G95

SiO₂ thermal growth: 1100°C, 100 mbar dry natural “¹⁶O₂”



+ vacuum annealing: 10⁻⁷ mbar, 700°C, 30 min

Samples preparation

Water partial pressure in air
of 30% humidity at 25°C

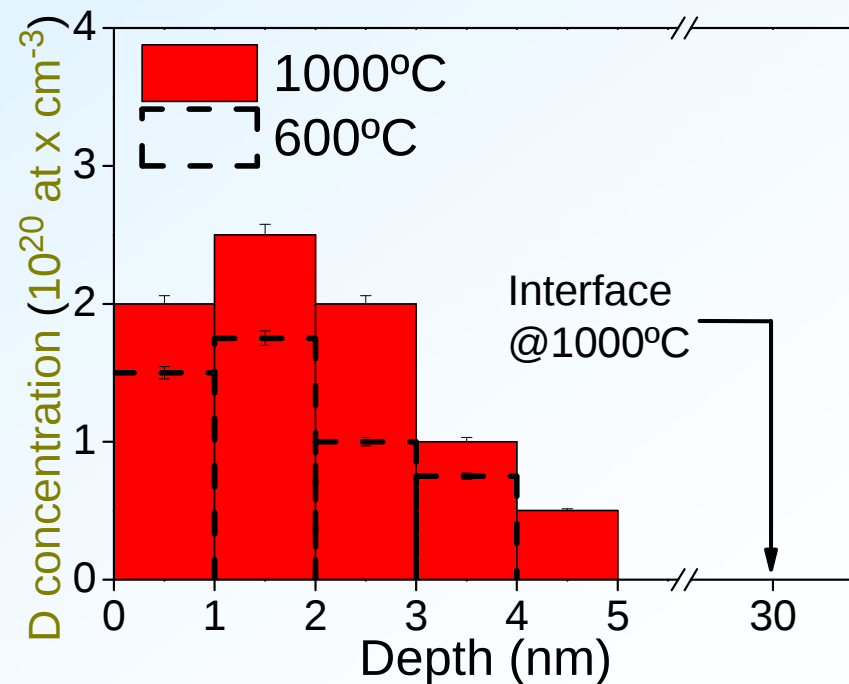
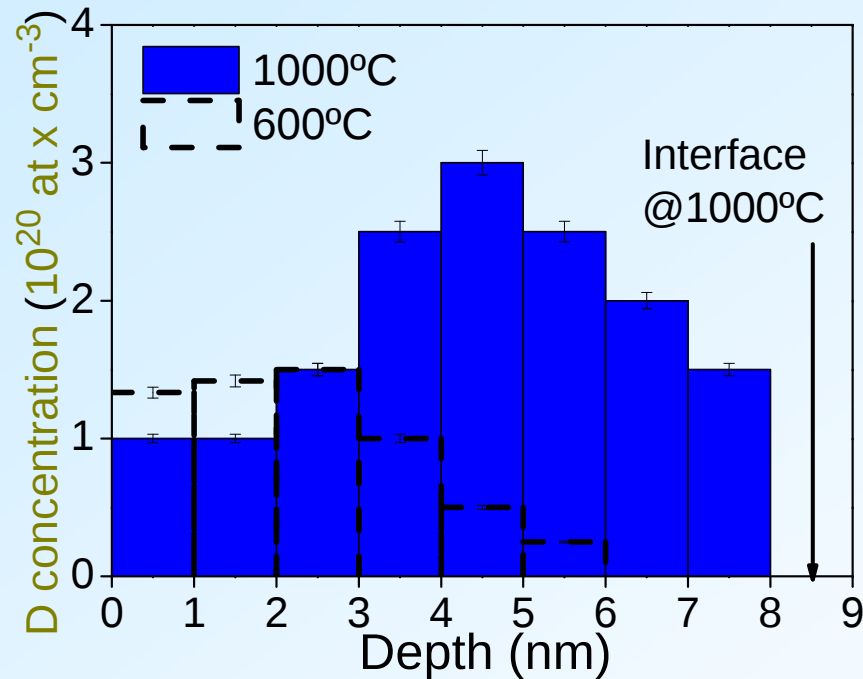
^{18}O – natural abundance of 0.2%
D – natural abundance of 0.015%

D_2^{18}O

Water vapor (D_2^{18}O) treatments: 1h, 200 – 800°C, 10 mbar

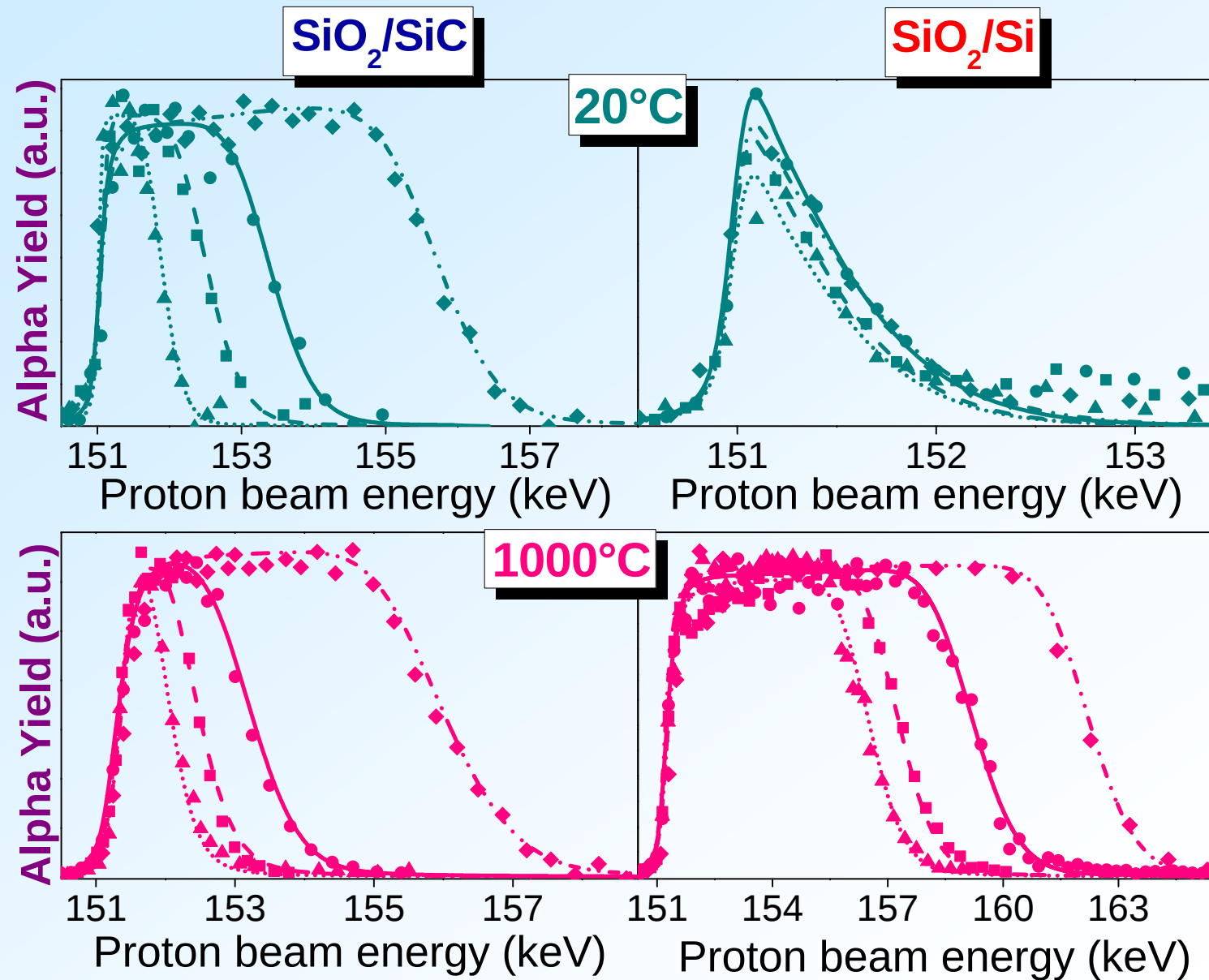
D profiles

in $\text{Si}^{16}\text{O}_2/\text{SiC}$ and in $\text{Si}^{16}\text{O}_2/\text{Si}$

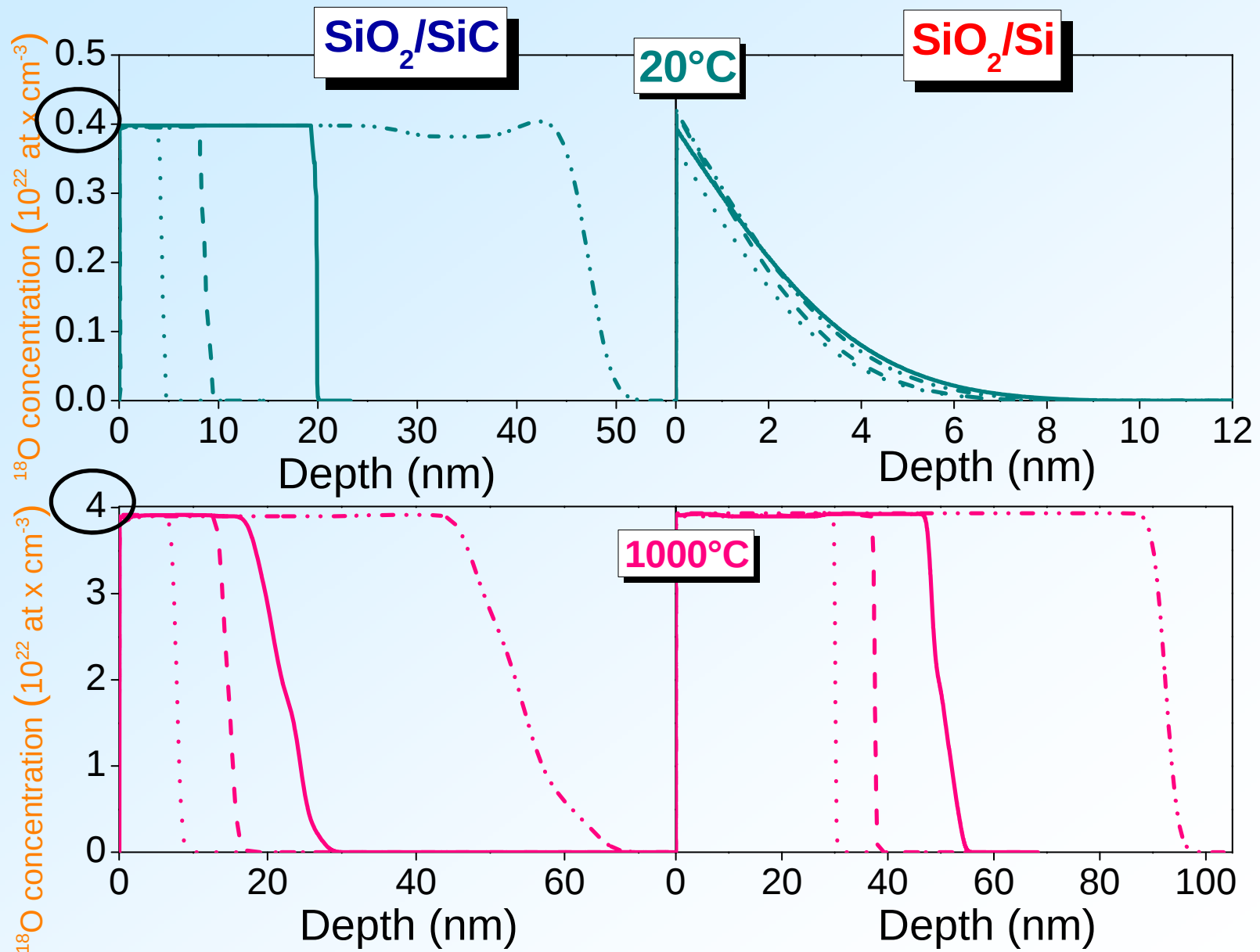


$\text{D}({}^3\text{He}, \text{p}){}^4\text{He}$ at 700 keV

^{18}O excitation curves



^{18}O profiles



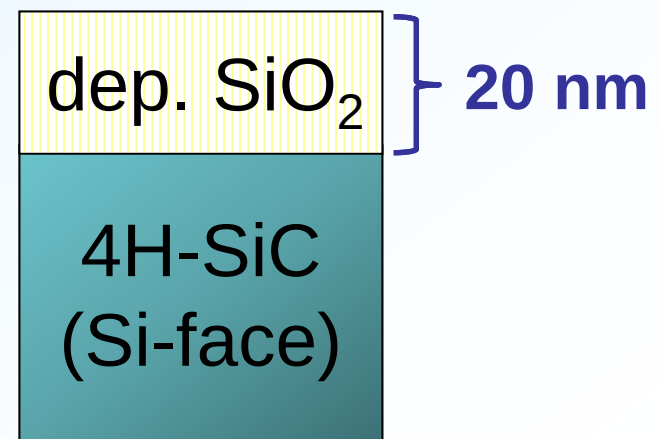
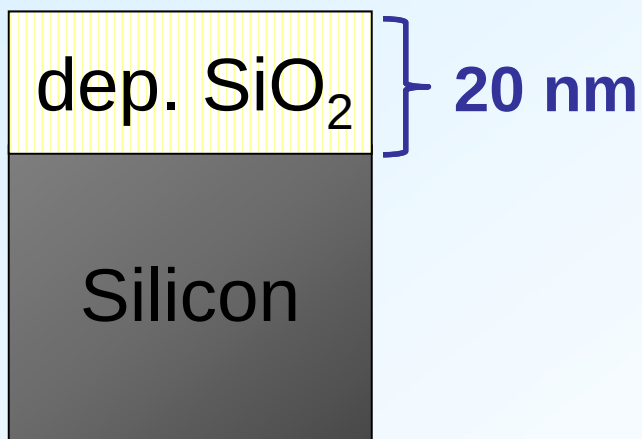
Experimental

Deposition on Si and on 4H-SiC substrates

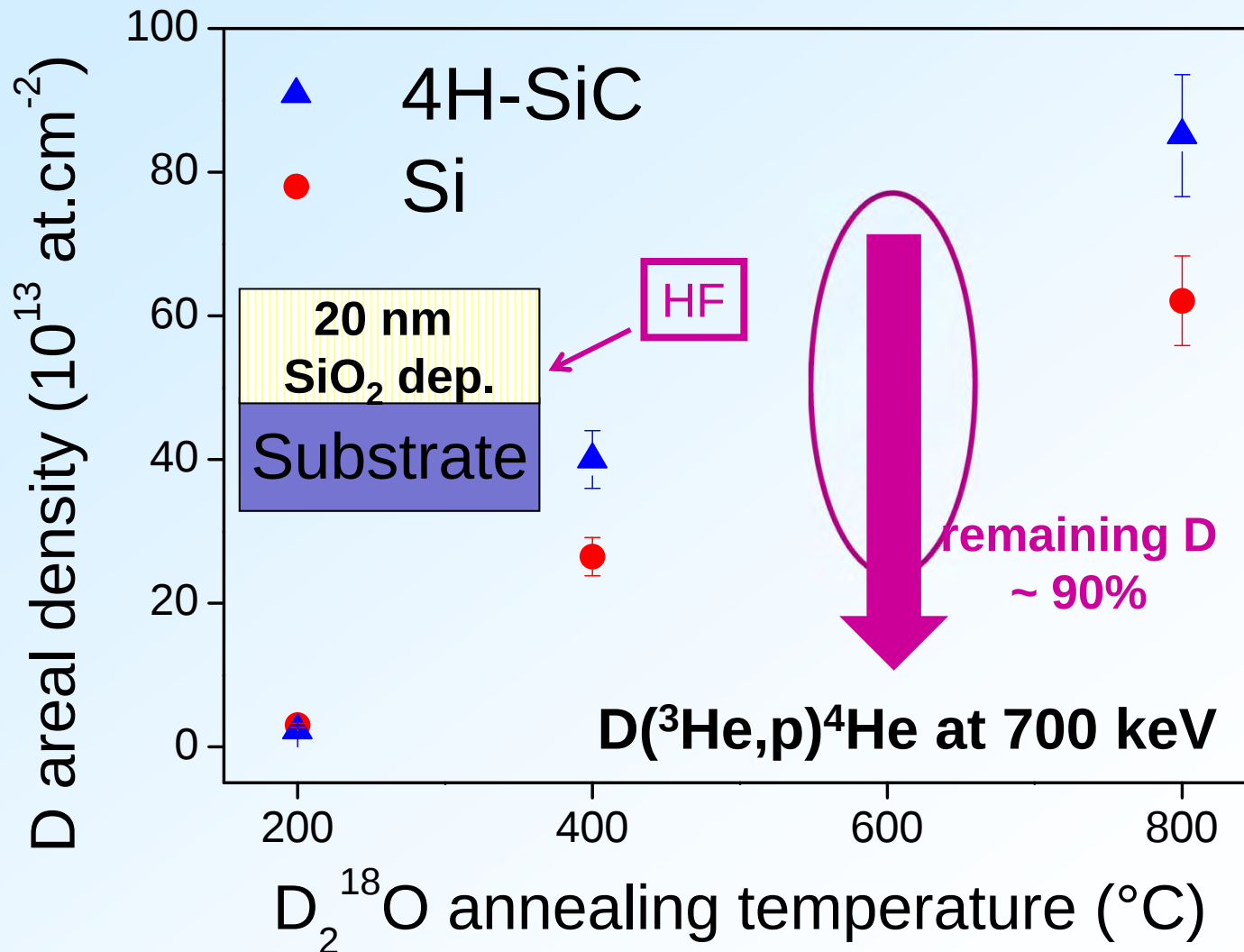
RF Sputtering conditions:

- Target: SiO_2 , 90 W
- 2 mtorr of Ar, flux of 20 sccm

Deposition rate $\sim 0.1 \text{ \AA/s}$
 $t = 2,000 \text{ s}$

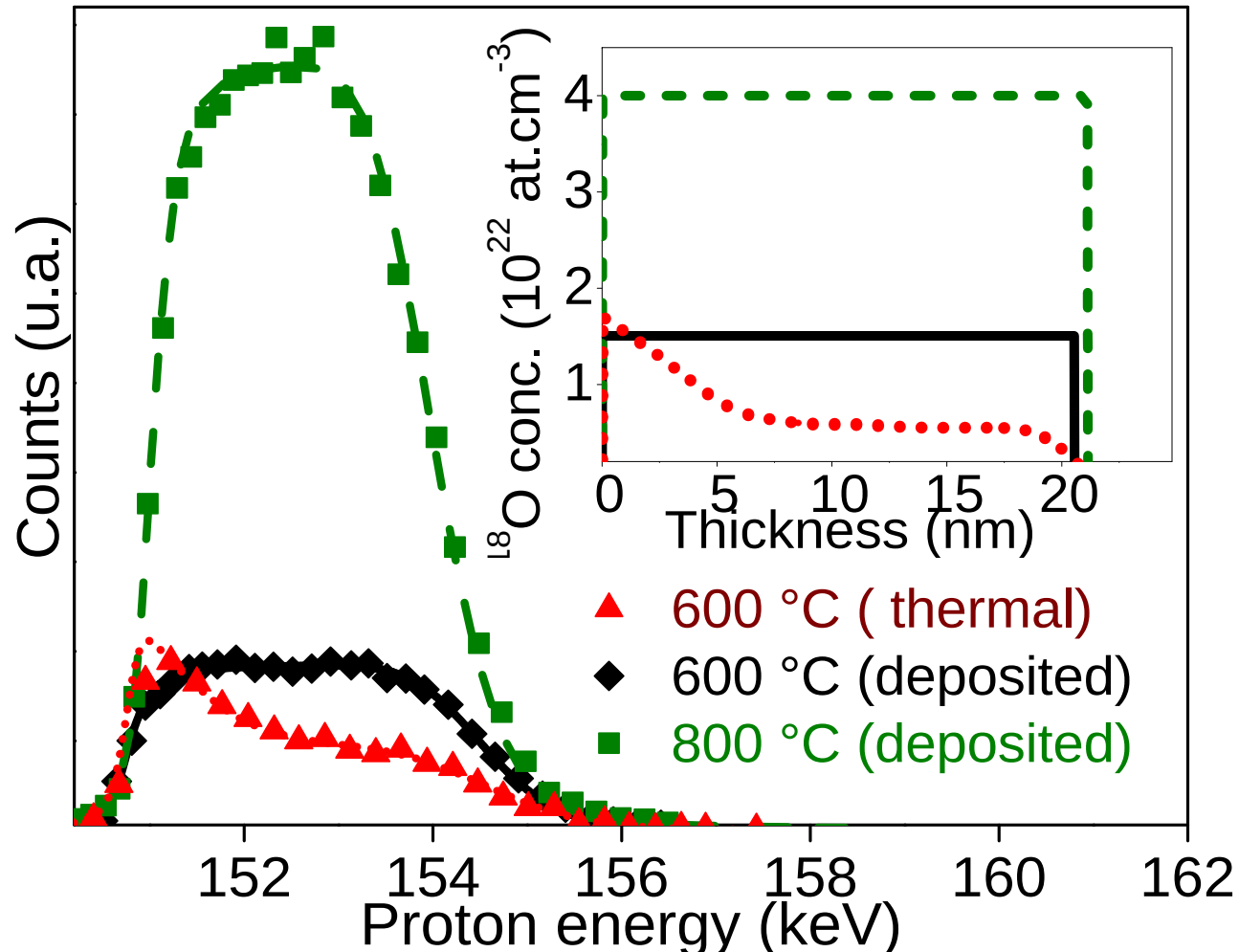


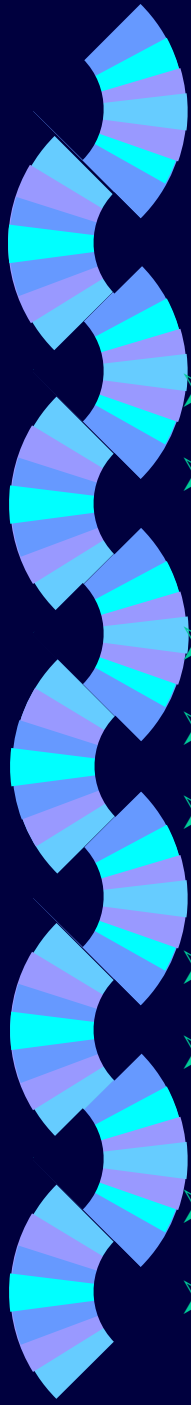
Temperature and substrate influence in $D_2^{18}O$ vapor incorporation



^{18}O profiles

SiO_2/SiC + 10 mbar D_2^{18}O , 1h

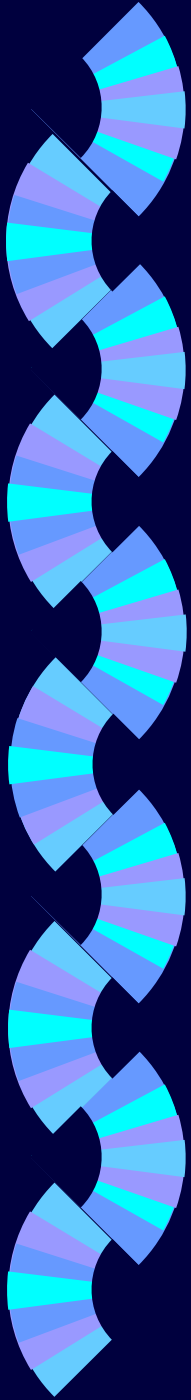


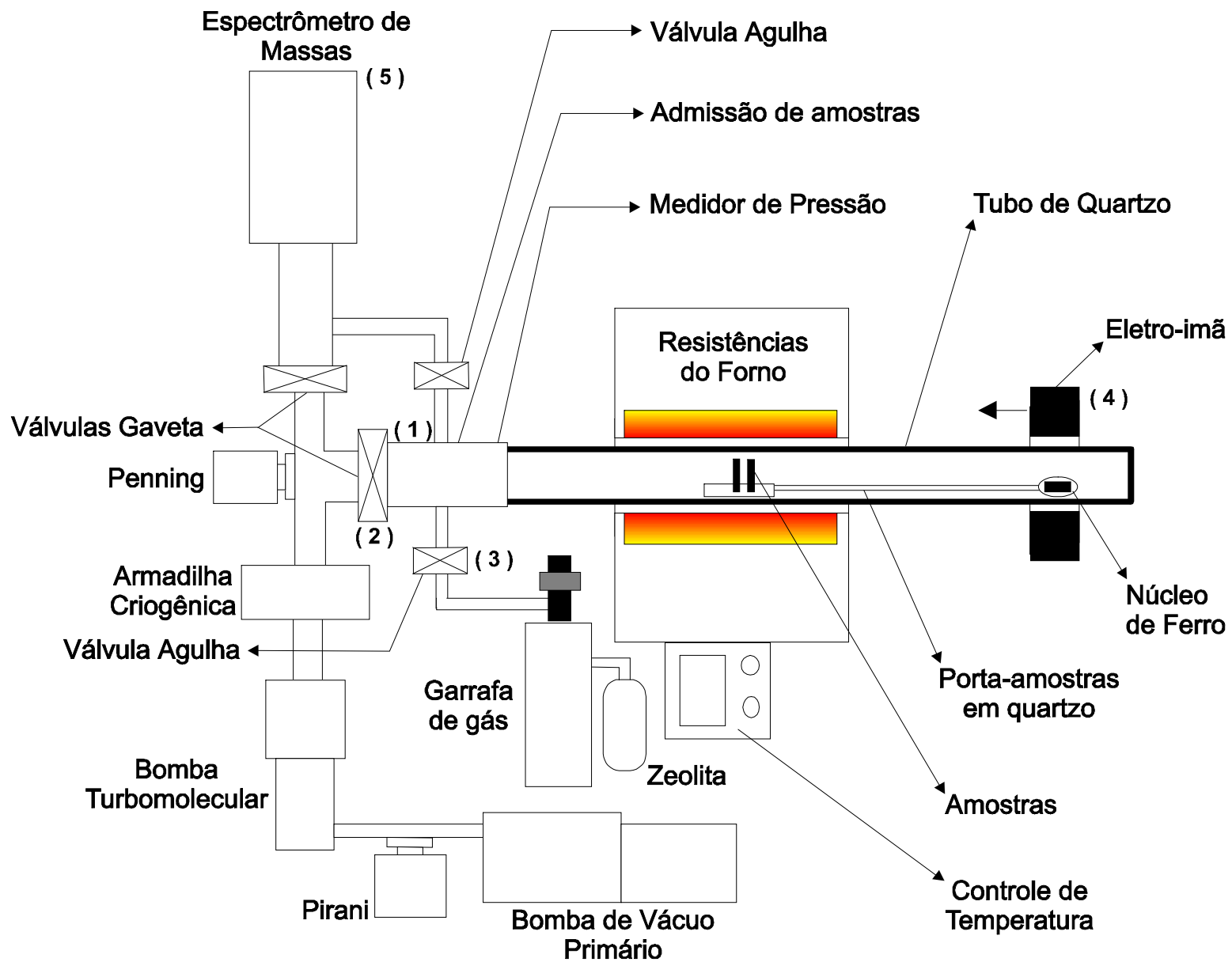


Further research topics

- Mechanism and limiting step of thermal growth of SiO_2 on SiC
- Annealings in D_2 of SiO_2 / 4H-SiC and SiO_2 / 6H-SiC structures with and without Pt
- Annealings in NO of Pt / SiO_2 / SiC structures
- Reoxidations and sequential annealings in H_2O_2
- Oxidations in $^{18}\text{O}_2$ of 6H and 4H-SiC (Si and C-faces) varying P, t and T
- Brief thermal growth followed by SiO_2 deposition
- Nitridation in $^{15}\text{NH}_3$ of SiC and of SiO_2 / SiC followed by SiO_2 deposition
- CO annealings of SiO_2 / SiC and SiO_2 / Si structures
- Incorporation and quantification of P in SiO_2 / SiC

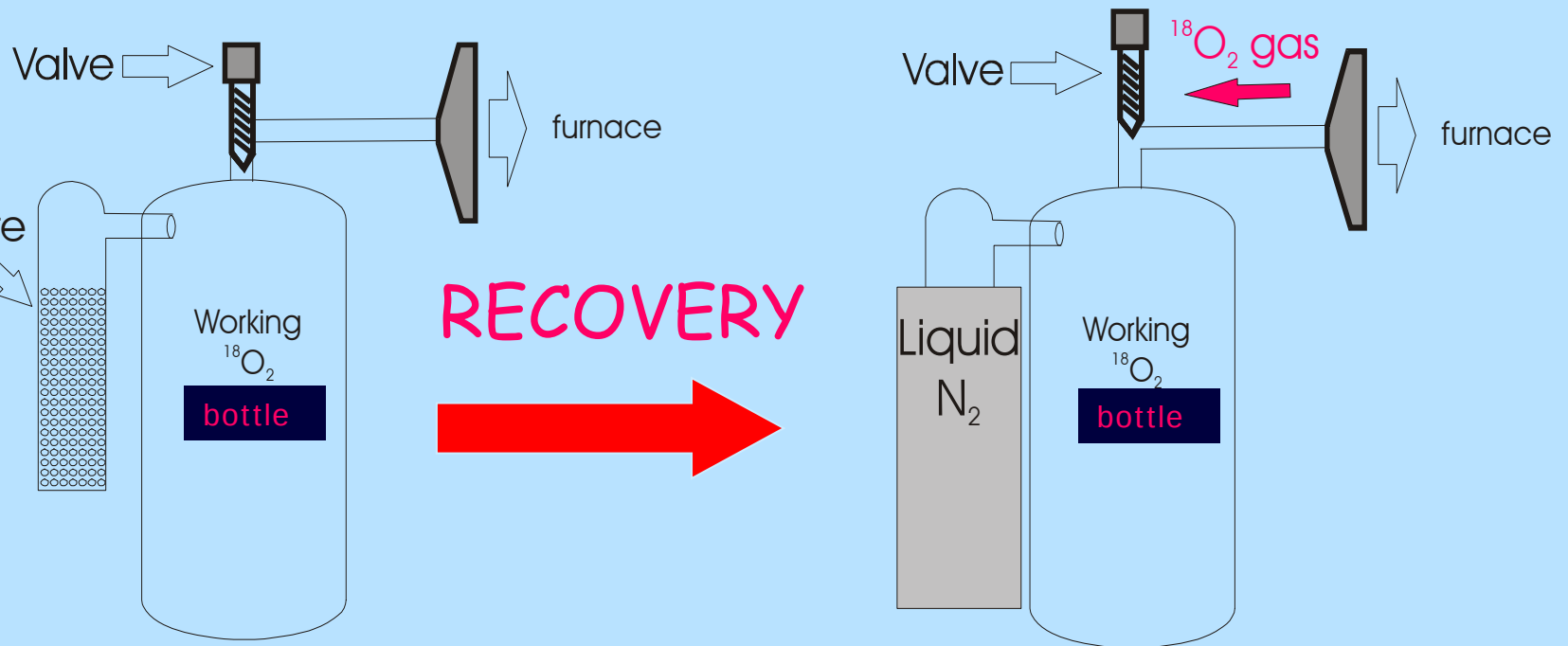
Thank you!





Manipulation and Recovering of $^{18}\text{O}_2$

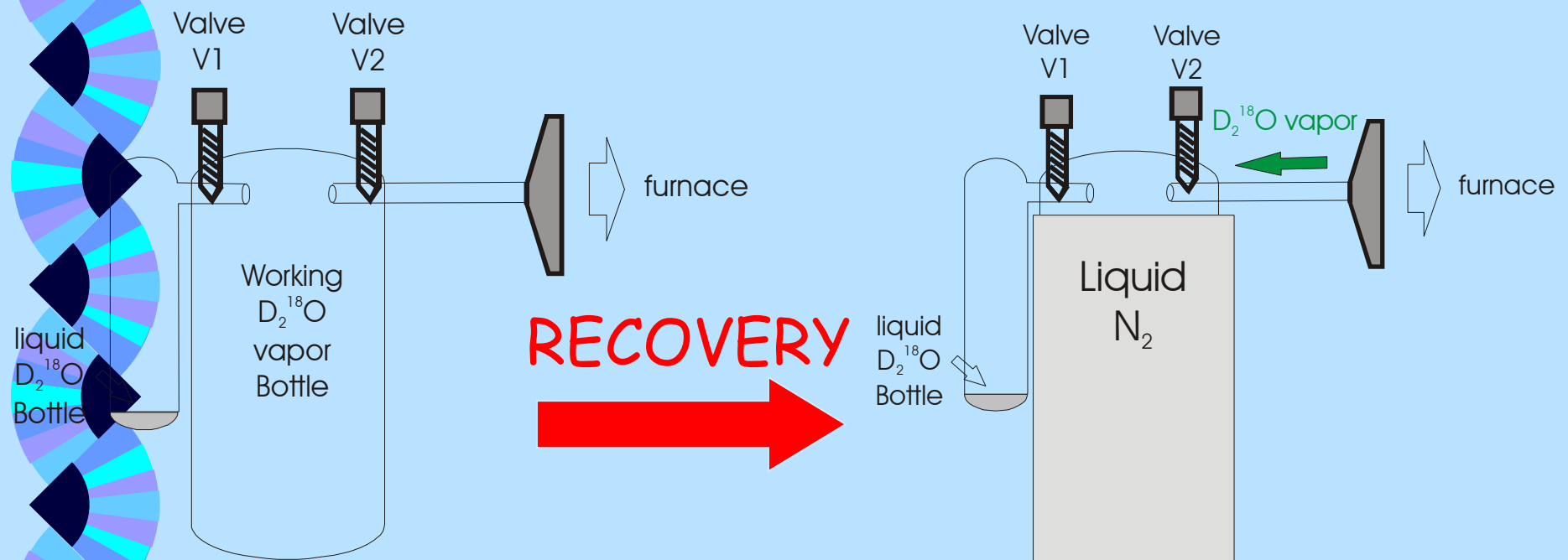
- Expensive gas: about U\$ 1,000/L
(97% enriched in the ^{18}O isotope)

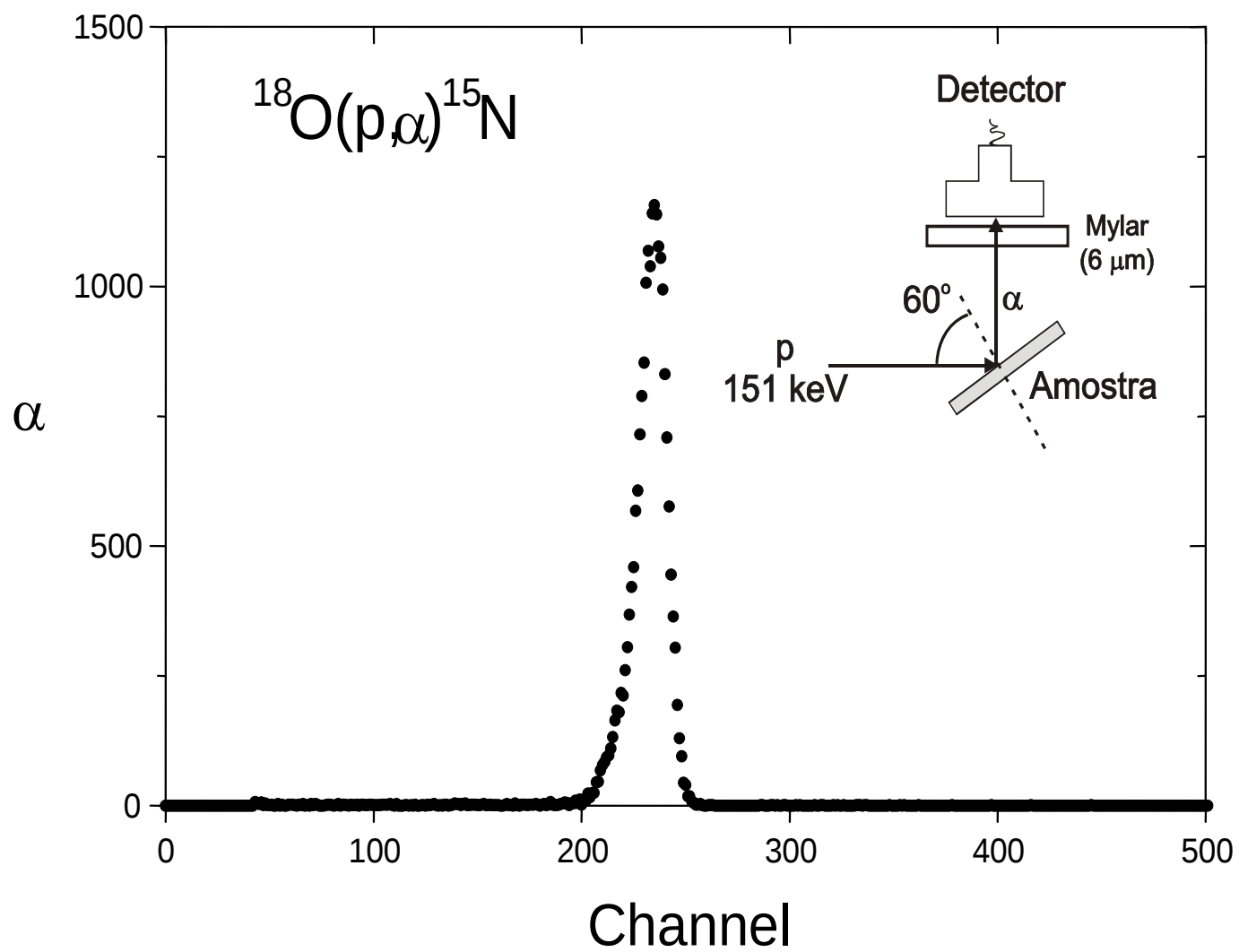
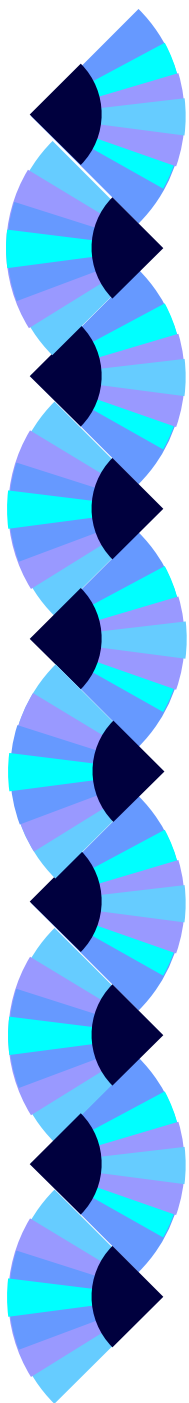


Natural Oxygen: 99.759% ^{16}O 0.204% ^{18}O 0.037% ^{17}O

Manipulation and Recovering of $D_2^{18}O$

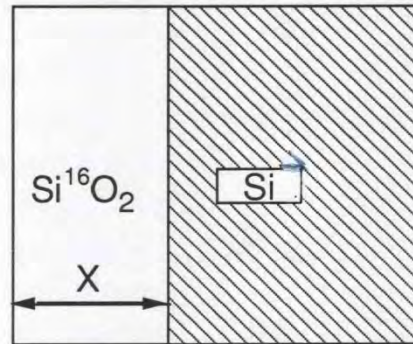
➤ VERY expensive : about U\$ 2,000/mL !!!



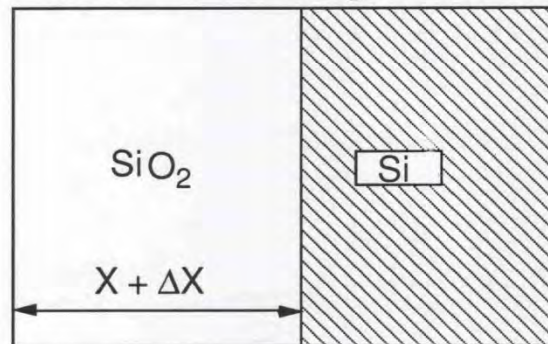


PRINCIPLES OF ^{18}O LABELING

1) oxidation in $^{16}\text{O}_2$ gas



2) oxidation in $^{18}\text{O}_2$ gas



determination of ^{18}O and ^{16}O profiles
and

comparison with theoretical profiles corresponding
to various growth mechanisms

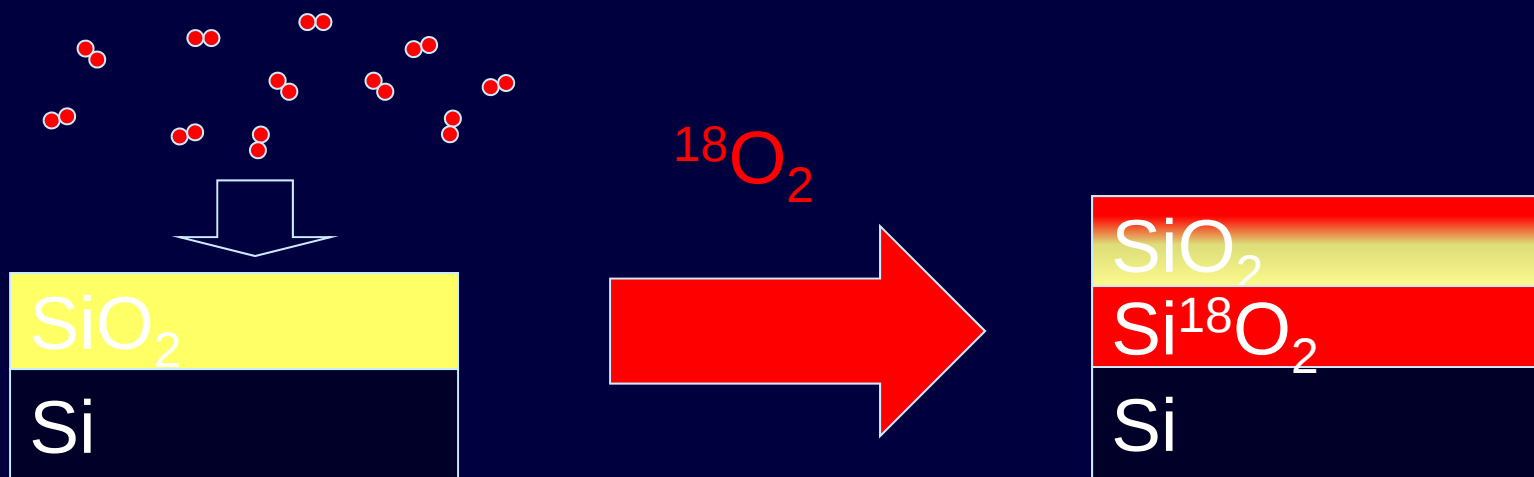
O_2 natural:
99,759% ^{16}O
0,204% ^{18}O
0,037% ^{17}O

$^{18}\text{O}_2$ 97%

Utilizando NRP para compreender o transporte atômico durante a oxidação do Si

Espécies móveis durante a oxidação?

Resultados:

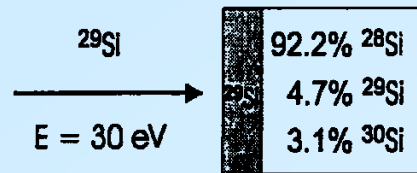


oxigênio é a espécie móvel, e que se desloca de duas maneiras:

- Por difusão, interagindo com os defeitos de rede (região superficial)
- Intersticialmente, sem reagir com o óxido já formado até chegar à interface com o semiconductor e reagir formando SiO_2

Isotopic tracing of Si in thermal growth of silicon oxide films on Si in dry O₂

Step 1: ²⁹Si deposition on Si (111)



Step 2: epitaxial recrystallization: 600 °C , 30 min , UHV

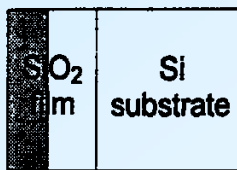
Step 3: ²⁹Si depth profiling

Step 4: thermal oxidation in O₂: 1000 °C , 60 min , 50 mbar

Step 5: ²⁹Si depth profiling

Step 6: comparison of ²⁹Si profile with predictions:

Si immobile and O mobile



Si mobile and O immobile

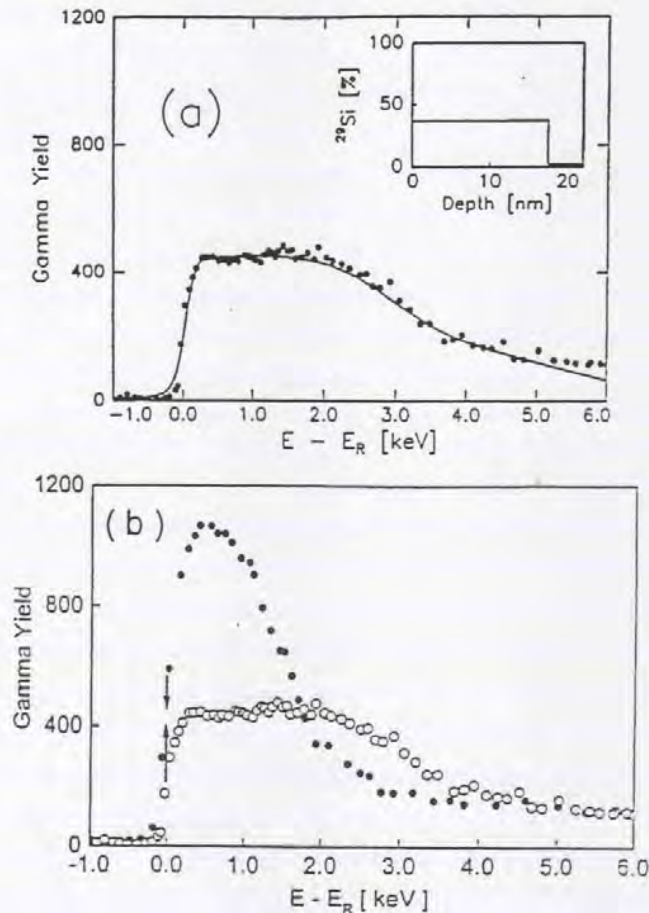
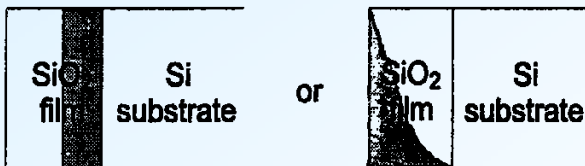
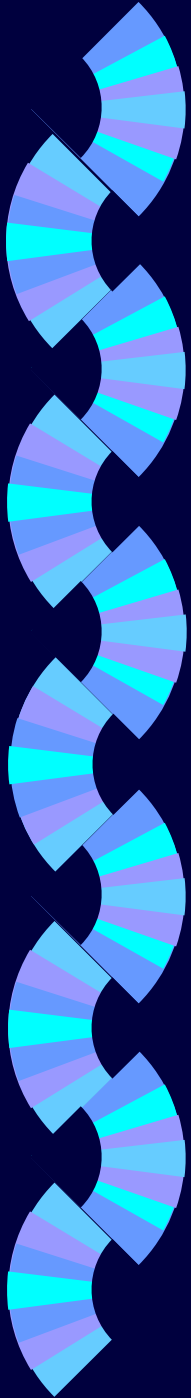
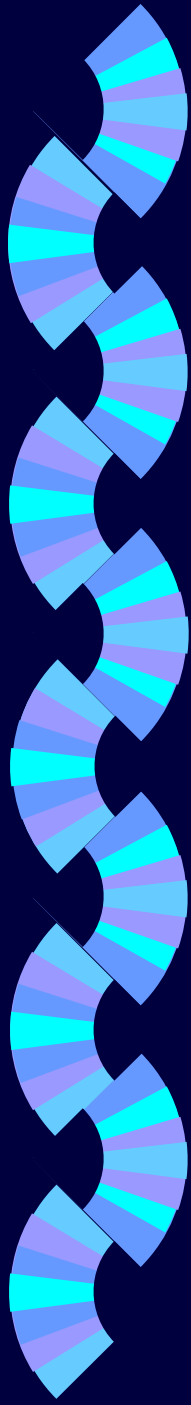


FIG. 4. (a) ²⁹Si excitation curve (solid circles), its simulation (solid line), and ²⁹Si profile in the inset for the oxidized sample. (b) ²⁹Si excitation curves in the pre-oxidation [solid circles, ²⁹Si epitaxial on Si(111)] and post-oxidation [empty circles, ²⁹Si in SiO₂/Si(111)] samples. The arrows indicate the energy positions of the half maxima in the leading edges for the pre-oxidation (arrow pointing downwards) and post-oxidation (arrow pointing upwards) samples.

I.J.R. Baumvol, C. Krug, **F.C. Stedile**, F. Gorris, W.H. Schulte
 Phys. Rev. B 60 (1999) 1492

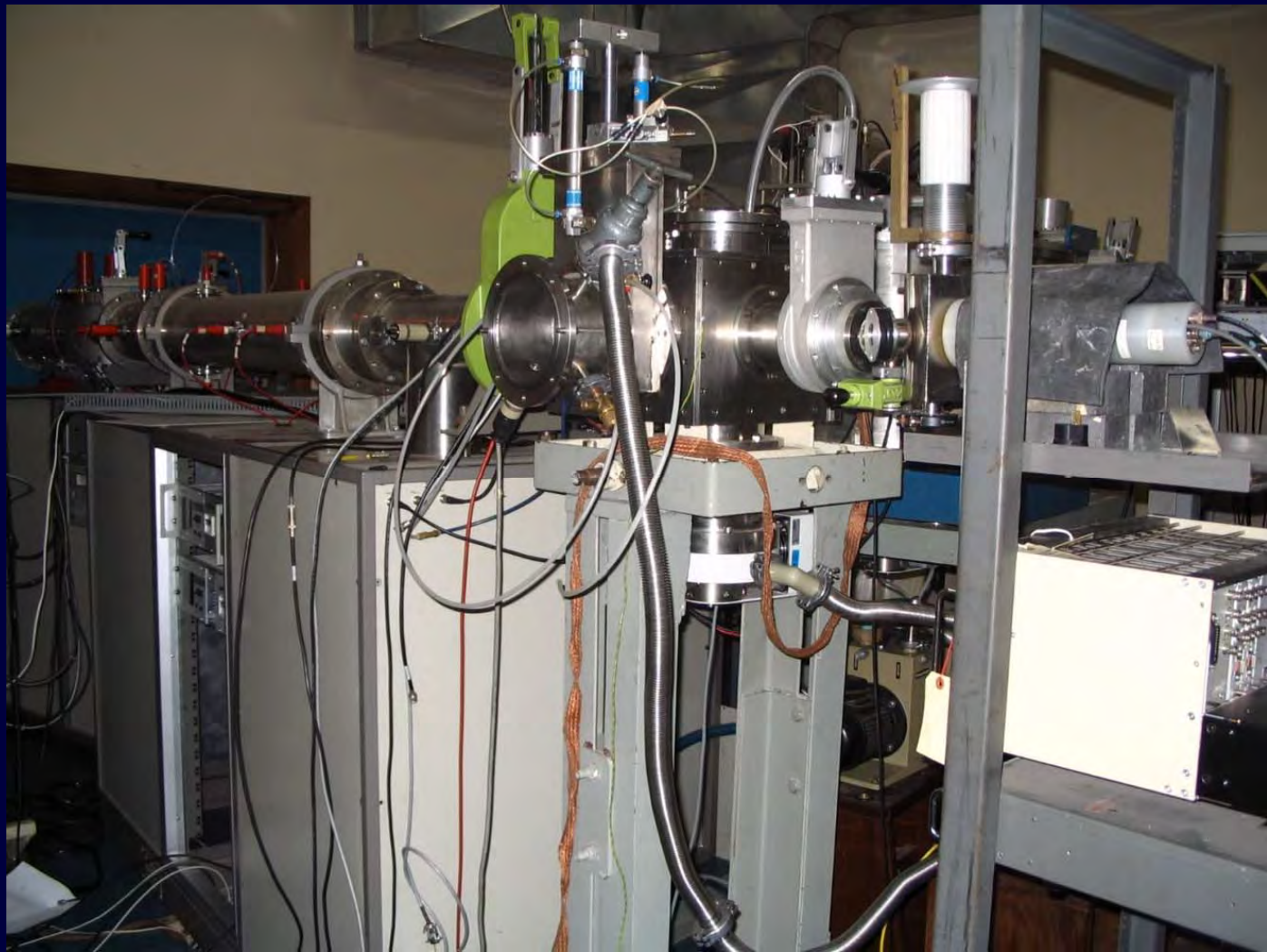
Ions decelerator



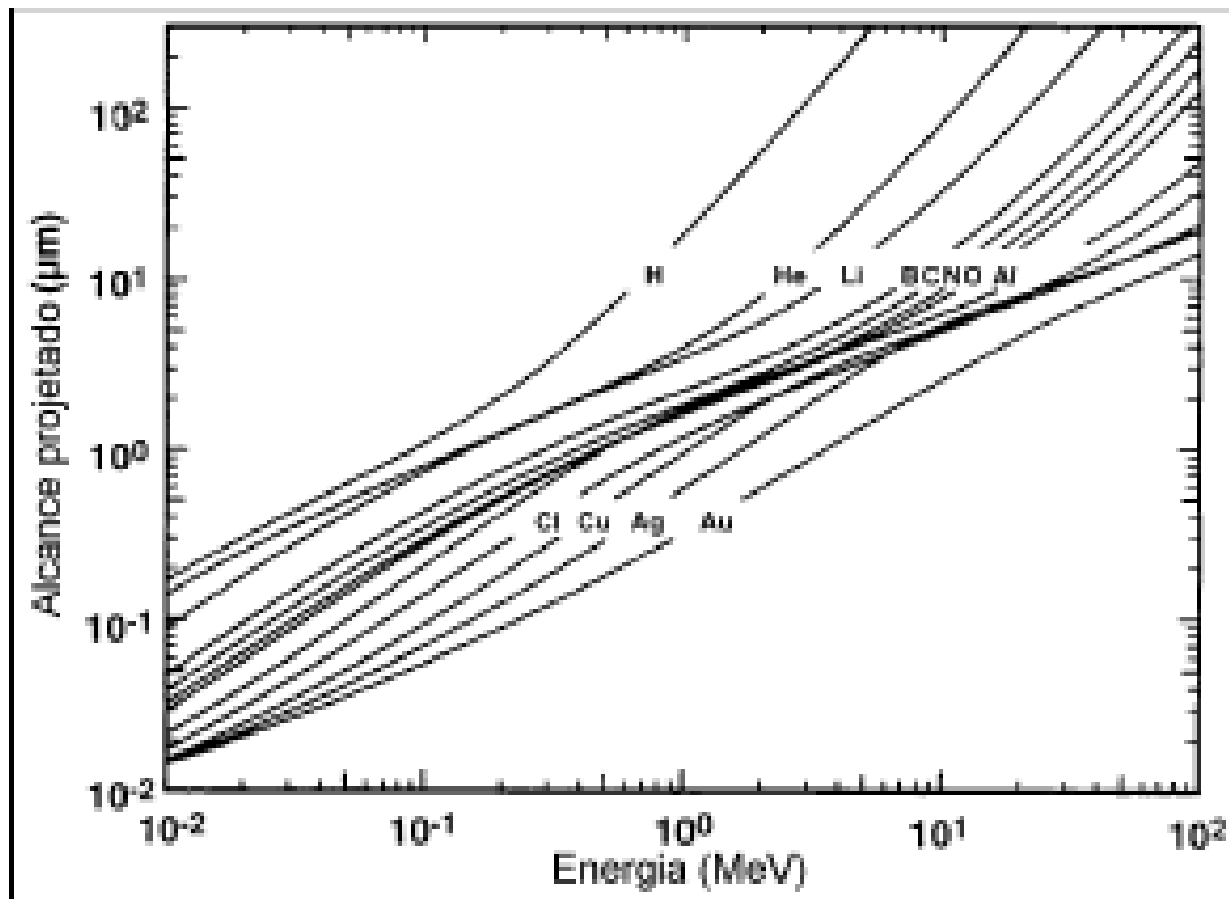


$^{29}\text{Si}(p, \gamma)^{30}\text{P}$

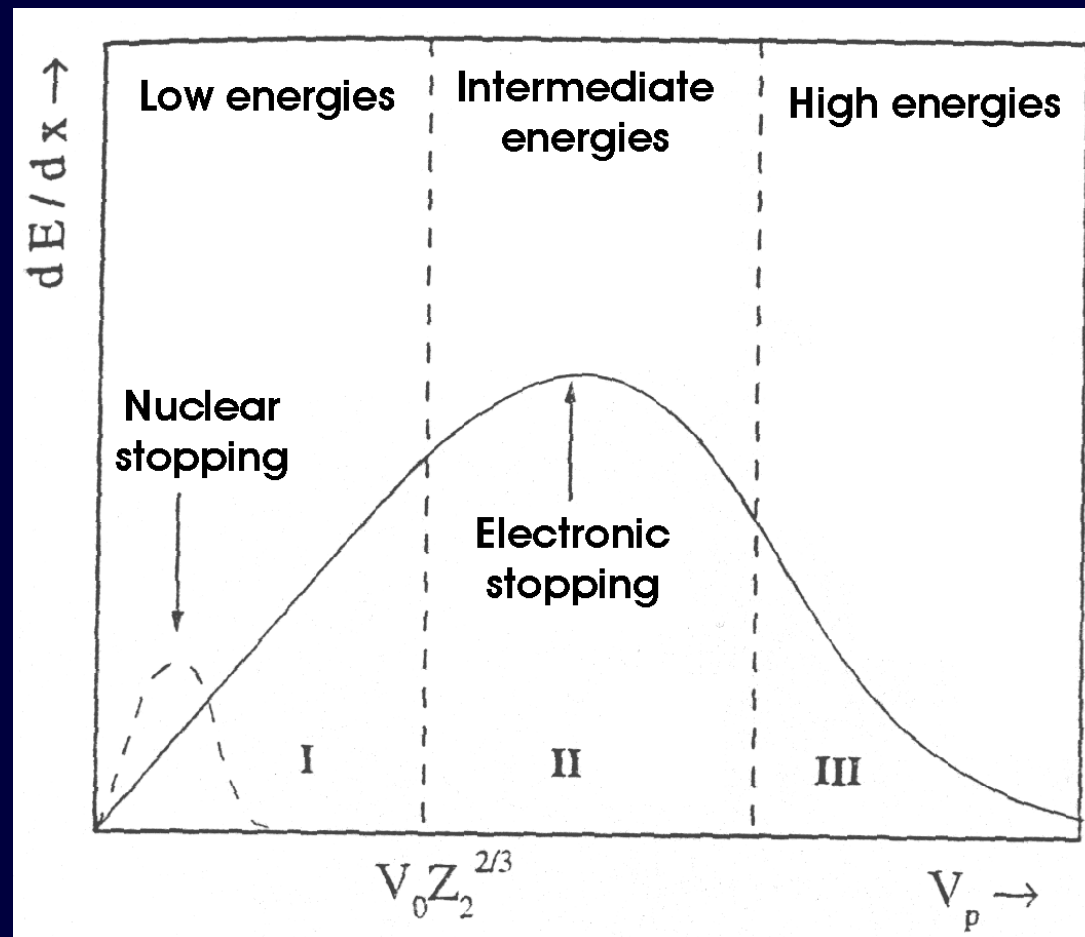
$E_R = 417 \text{ keV}; \Gamma = 100 \text{ eV}$



Alcance das partículas no Mylar

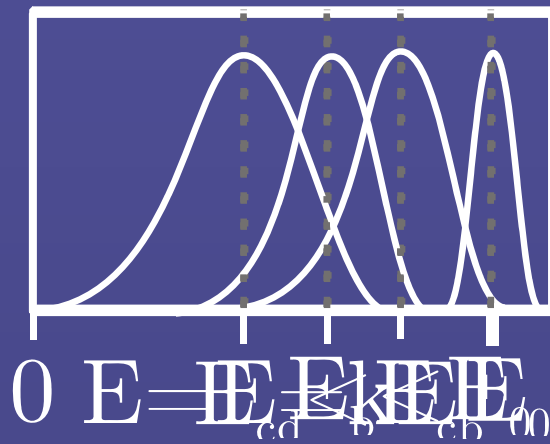
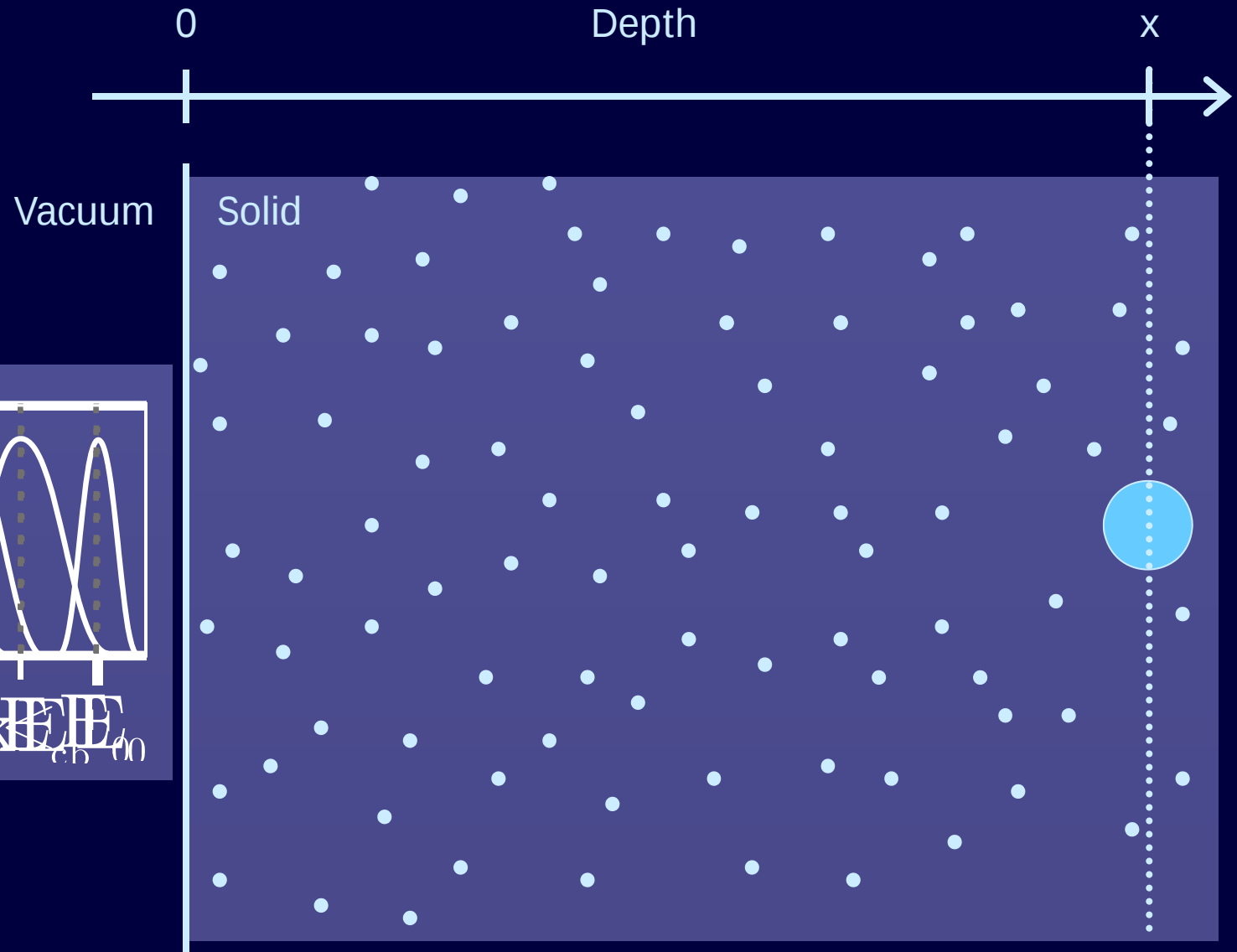


Electronic and nuclear dE/dx

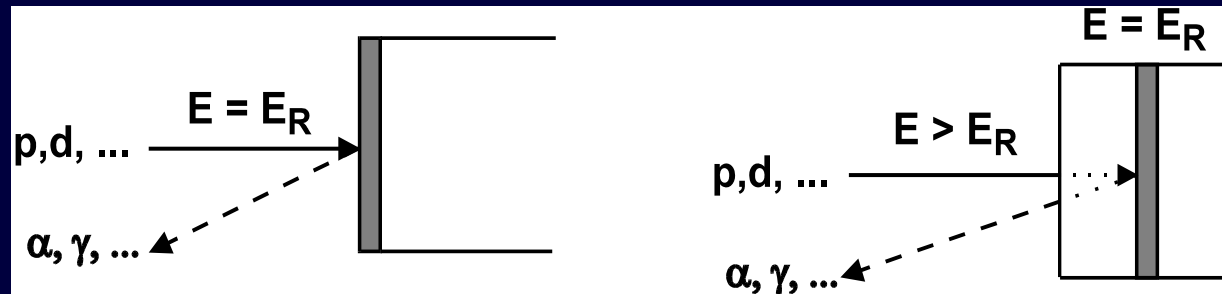


Energy loss

$$k_i = \left[\frac{\sqrt{m_t^2 + m_i^2 \sin^2 \theta} + m_i \cos \theta}{m_i + m_t} \right]^2$$



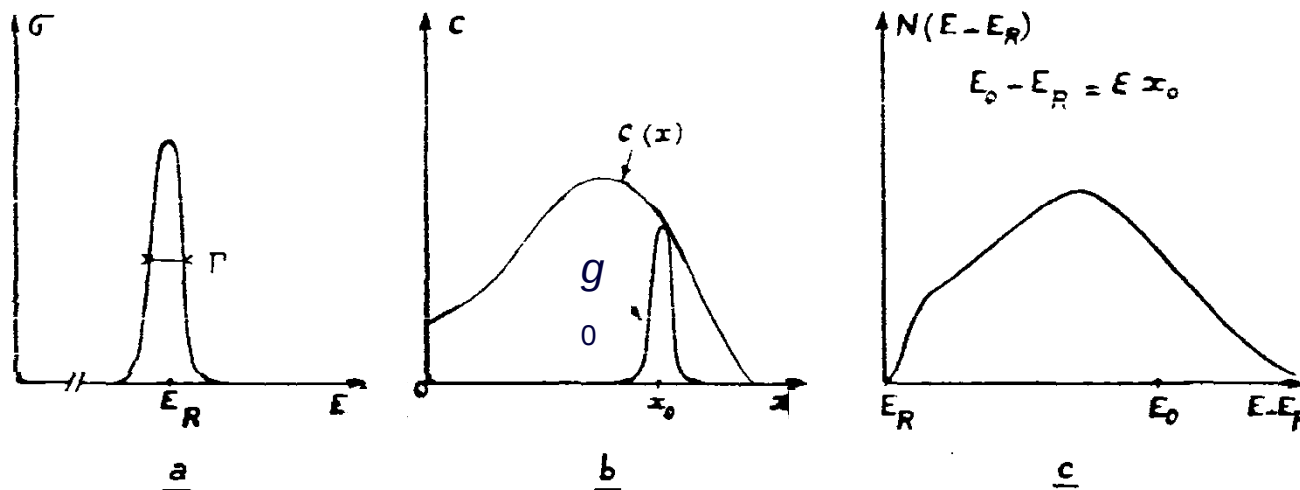
Narrow Resonance Nuclear Reaction Profiling: NRP



Isolated Resonance

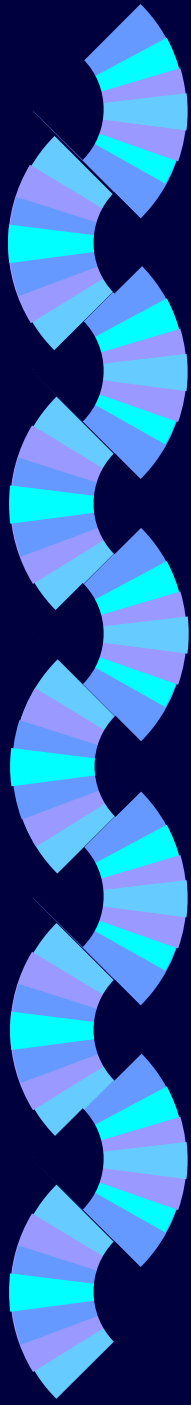
Concentration Curve

Excitation Curve



Interpretation: SPACES code

I. Vickridge and G. Amsel, Nucl. Instr. and Meth. B 45 (1990) 6



FLATUS

beam particle energy, mass and atomic number; energy loss per unit length and energy straggling corresponding to the material and energy of interest



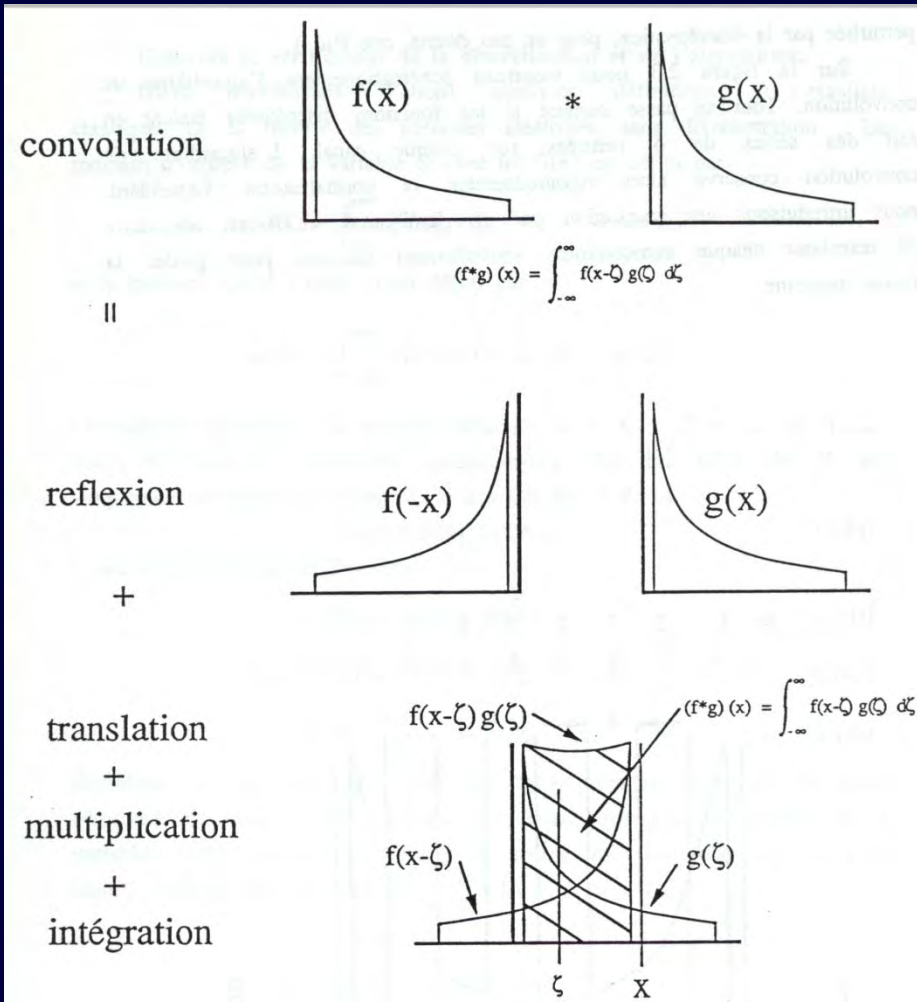
AUTOCONVOLUTION FILE

convolution with resonance line shape (assumed Lorentzian) and width, ion beam energy spread (assumed Gaussian) and its widening due to the Doppler effect at room temperature (assumed Gaussian), and concentration of the probed nuclide and thickness of each layer of a first tentative profile



THEORETICAL EXCITATION CURVE

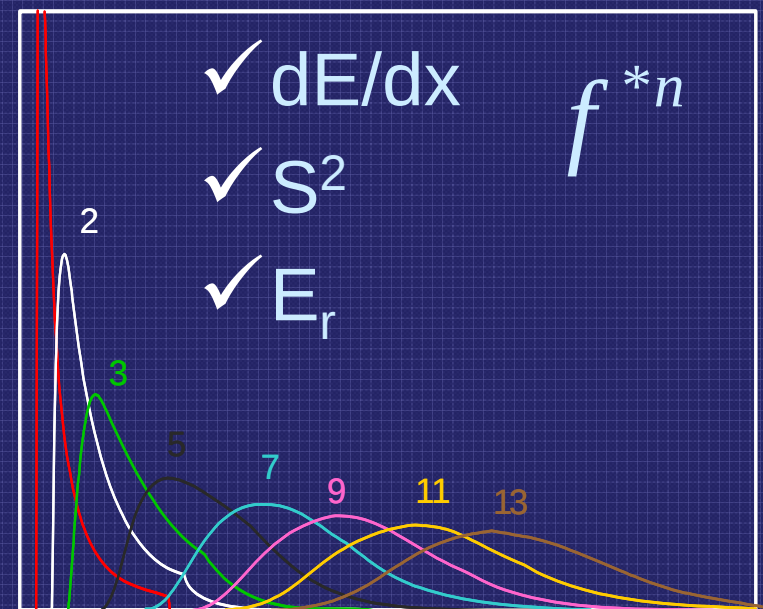
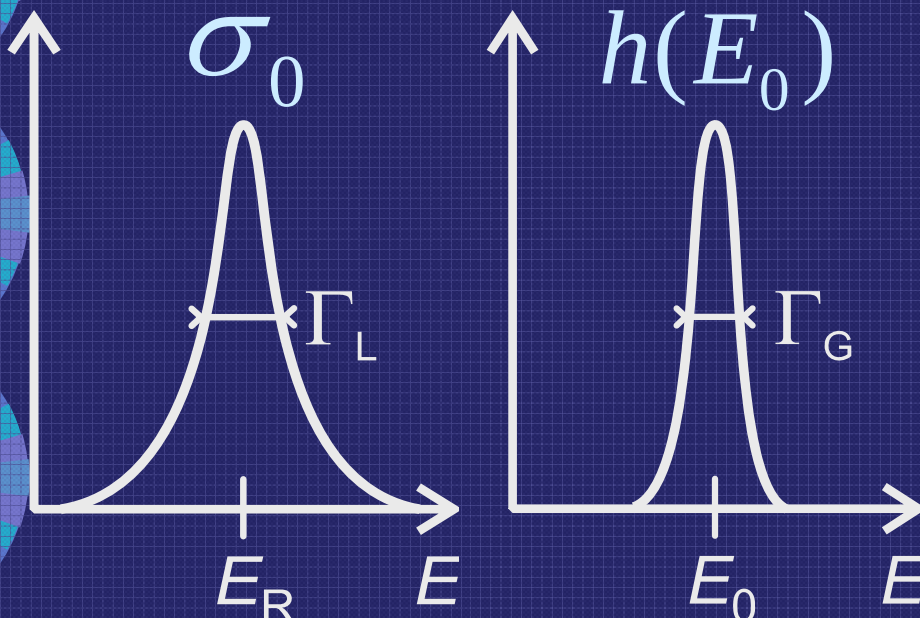
Convolution



I. Vickridge, Thesis, University of Paris VII (1990)

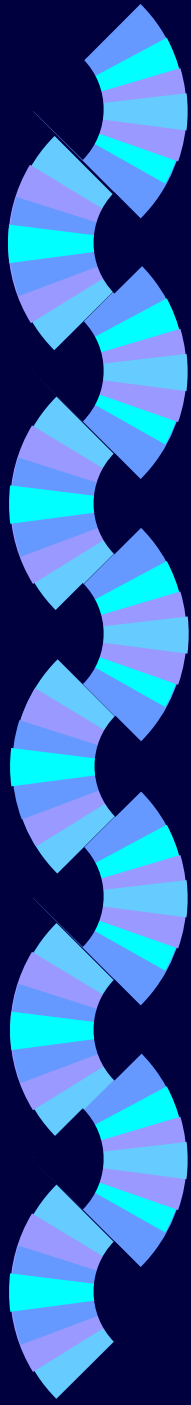
Modeling

$$N(E_0) = c \cdot \sigma_0(E) * h(E_0) * \sum_{n=0}^{\infty} K_n f^{*n}(E - E_0)$$



$$K_n = \int_0^{\infty} \frac{(mx)^n}{n!} e^{-mx} C(x) dx$$

$C(x) = ?$



Exemplo de Aplicação: ^{18}O em Si

1965: modelo de Deal e Grove p/ oxidação térmica do silício

Deal and Grove, JAP **36**, 3770 (1965)

❖ Principais suposições:

- existência de uma camada de óxido inicial (~ 20 nm);
- regime estacionário;
- difusão de O_2 através do filme de óxido de silício e reação na interface SiO_2/Si .

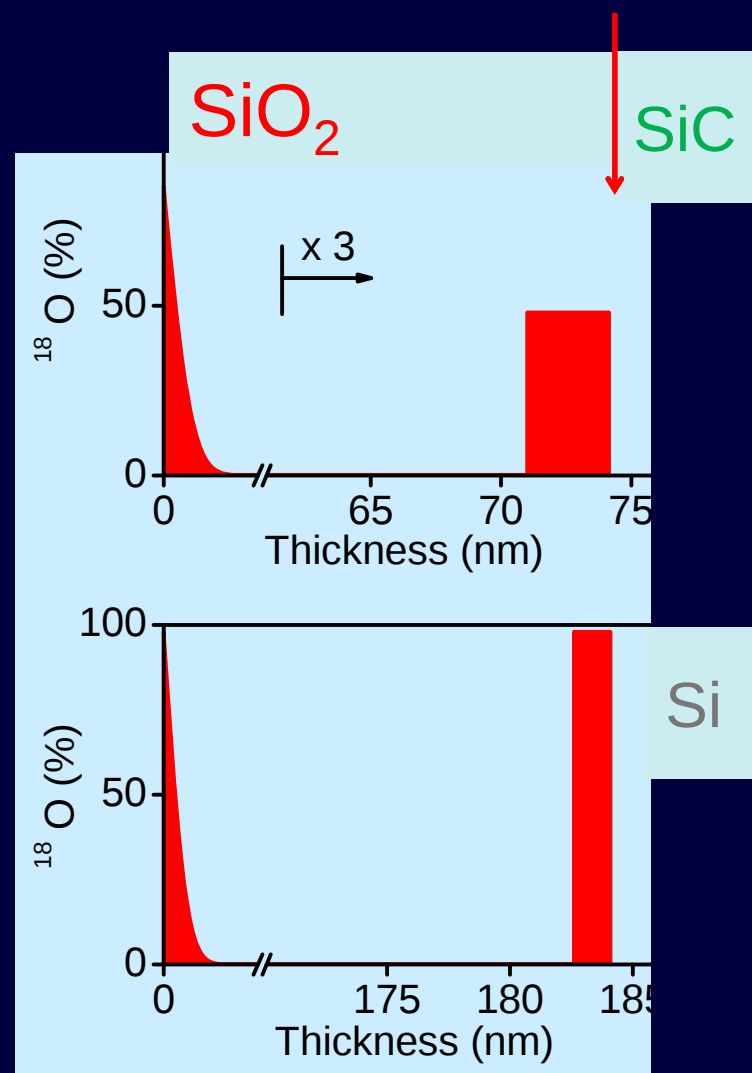


Osbourne Executive produzido nos anos 80 x iPhone em 2009:
 $28.75 \text{ pounds} / 135\text{g} = 100 \text{ x} + \text{pesado}$
 $4\text{MHz} / 412 \text{ MHz} = 100 \text{ x} + \text{lento}$
 $\$2500 / \$200\text{-}300 = 10 \text{ x} + \text{caro}$
 $(52\text{cm} \times 23\text{cm} \times 33\text{cm}) / (115\text{mm} \times 61\text{mm} \times 11.6\text{mm}) = 485 \text{ x maior}$

Mecanismo e Etapa Limitante do Crescimento Térmico de SiO_2 sobre SiC

I.C. Vickridge *et al.*, Phys. Rev. Lett. 89 (2002) 256102

- Traçagem Isotópica – ^{16}O / ^{18}O
- $^{18}\text{O}(\text{p},\alpha)^{15}\text{N}$ $E_p = 151$ keV
- 6H-SiC tipo n polidas nas faces (0001) e (0001) e Si (001)



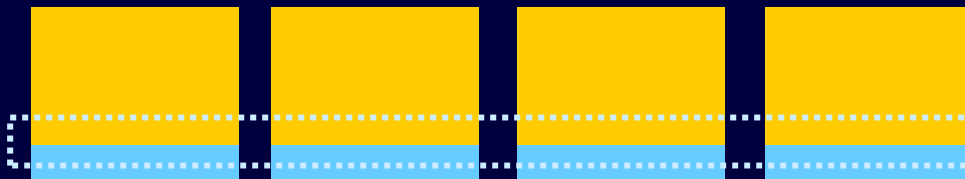
1º grupo de amostras

 Si^{16}O_2
 Si^{18}O_2
 SiC ou Si

 Limpeza das amostras

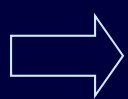
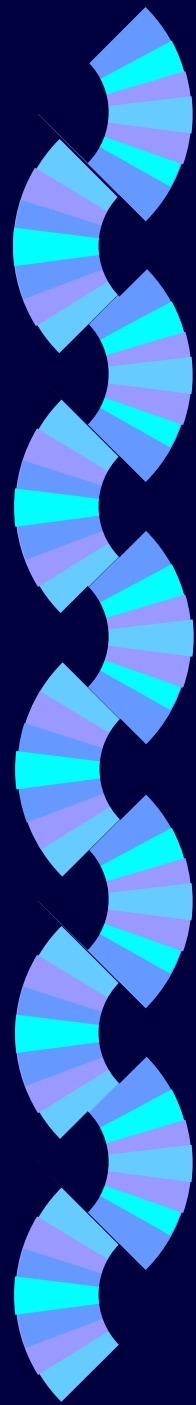


 $1100\text{ }^\circ\text{C}$ 100mbar $^{16}\text{O}_2$ $\frac{8\text{ h}}{45\text{ h}} \rightleftharpoons \text{Si}$
 $\rightleftharpoons \text{SiC}$



Interfaces
Idênticas





HF: taxa de ataque ~ 1 nm/s

70 s

40 s

30 s

1 s

$\Rightarrow$ Si

30 s

20 s

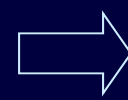
10 s

1 s

$\Rightarrow$ SiC



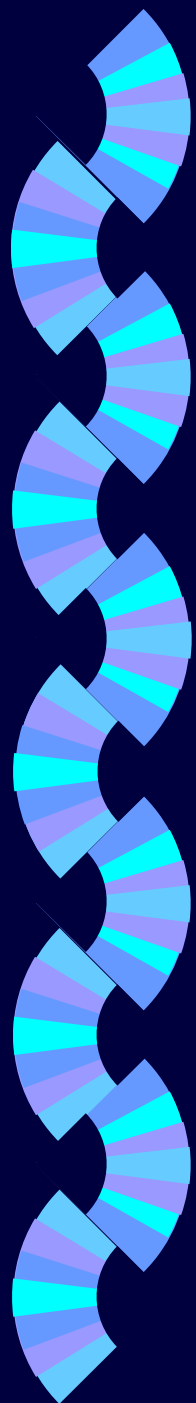
Interfaces
Idênticas



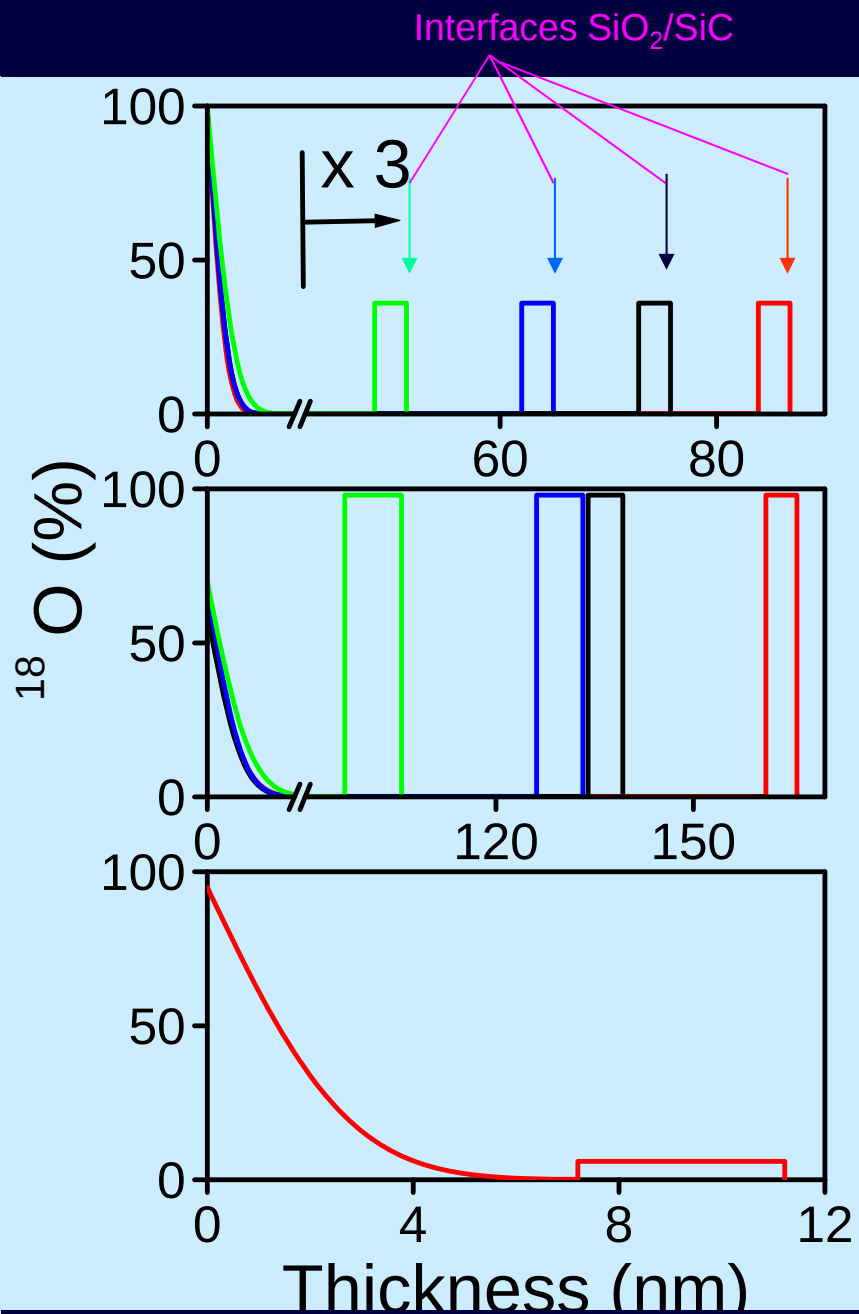
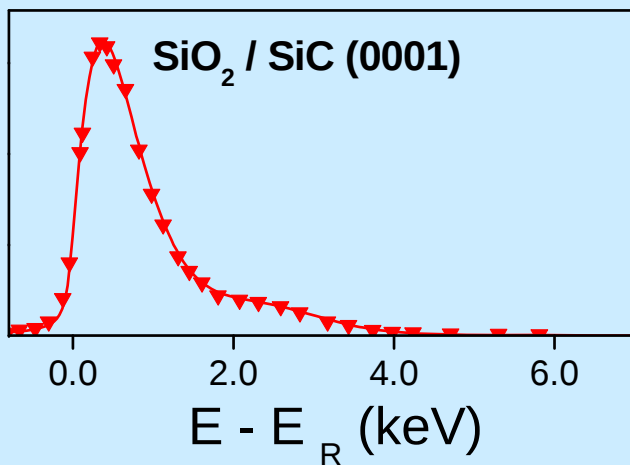
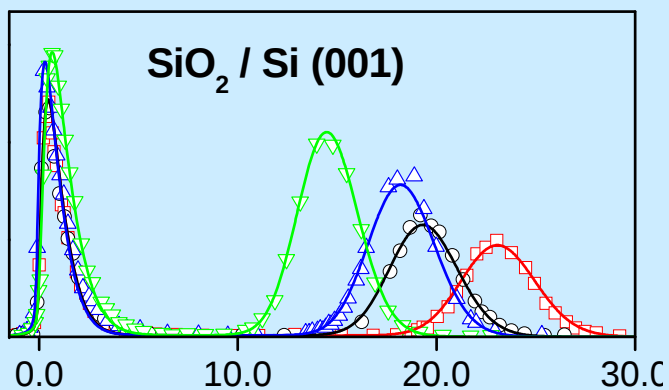
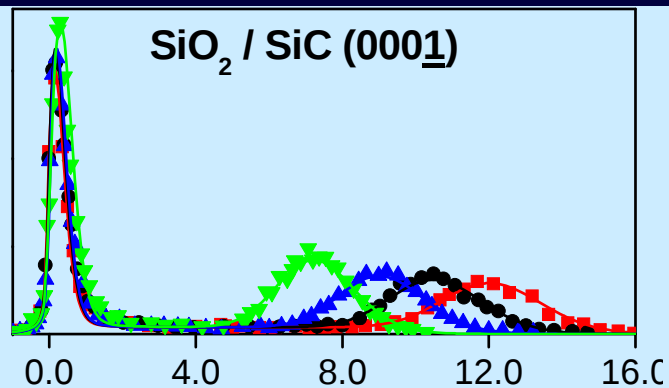
1100°C 100 mbar 1h $^{18}\text{O}_2$



 Si^{16}O_2
 Si^{18}O_2
 SiC or Si



Alpha Yield (arbitrary units)



2º grupo de amostras

Limpeza das lâminas



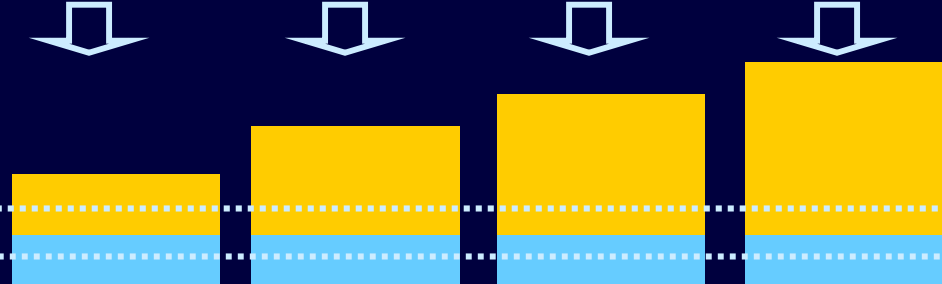
1100°C 100 mbar $^{16}\text{O}_2$

16 h

26 h

35 h

45 h



SiC ou Si

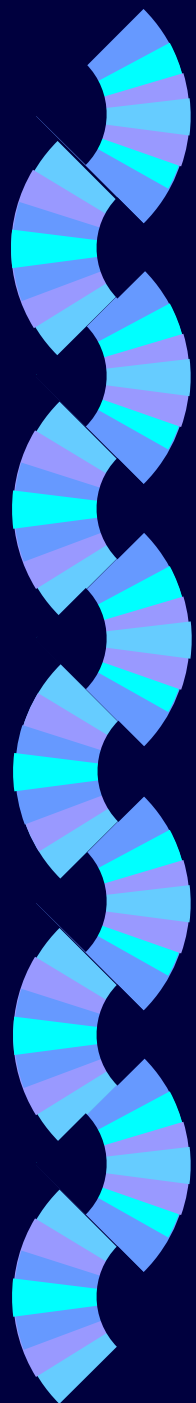
Si^{16}O_2

Si^{18}O_2

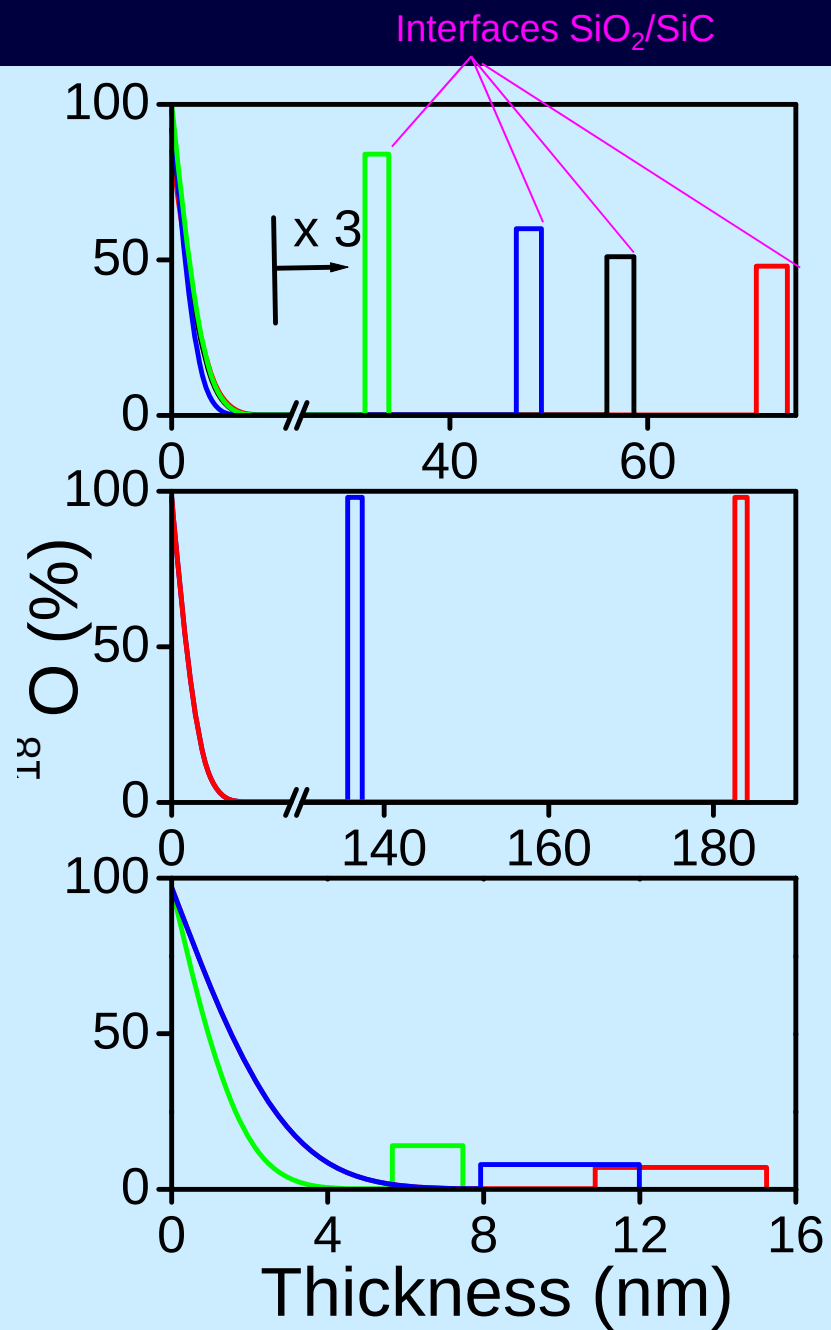
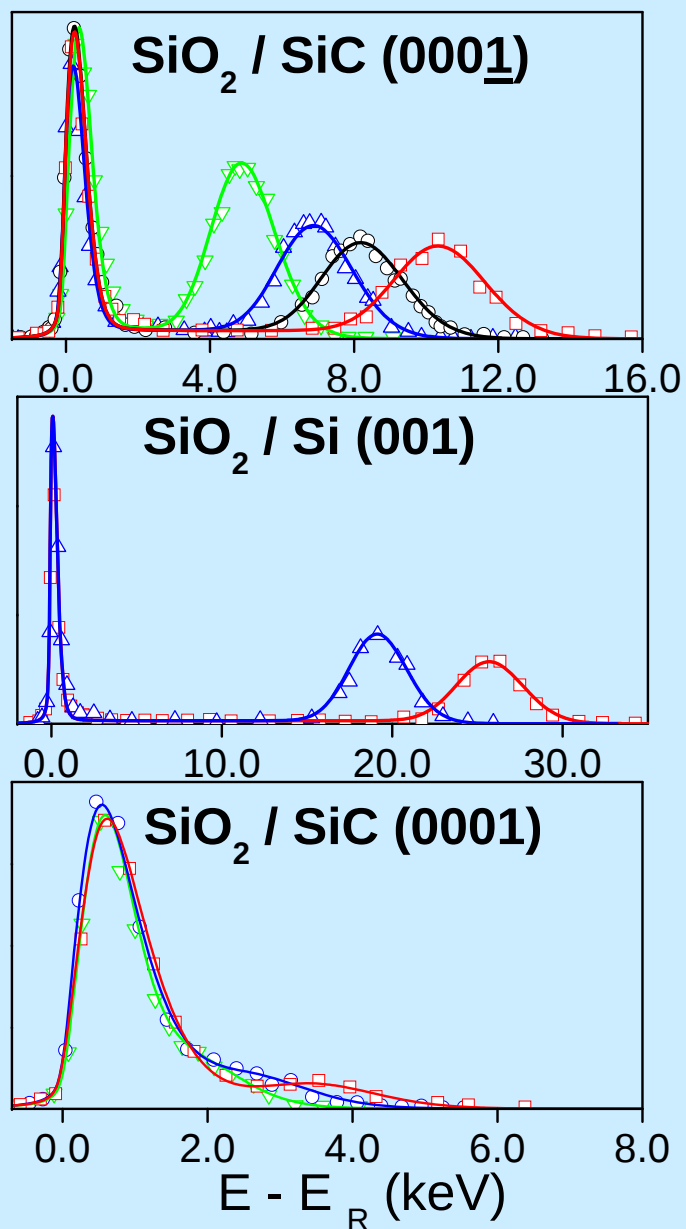
Interfaces distintas ?

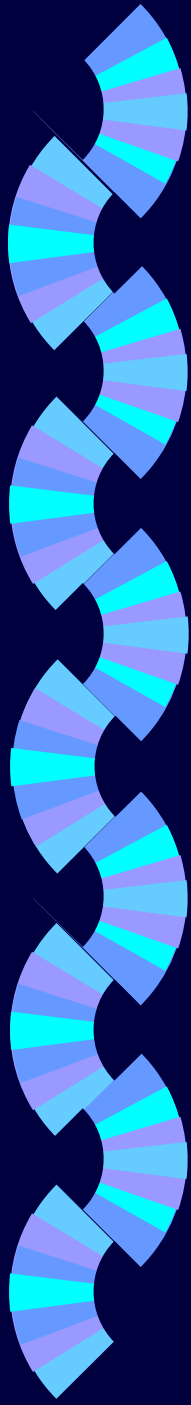
1100°C 100 mbar $^{18}\text{O}_2$ 1h





Alpha Yield (arbitrary units)





Colaboradores atuais

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- Profs. Gabriel V. Soares, Henry Boudinov – IF UFRGS
- Profs. Cláudio Radtke – IQ UFRGS
- Prof. Rodrigo Prioli Meneses – DF PUC-RJ
- Prof. Sima Dimitrijevic, Griffith University, Australia
- Prof. Leonard Feldman, Rutgers University, E.U.A.
- Dr. Aivars Lelys, Army Research Lab., E.U.A.
- Dr. Anant Agarwal, Cree Inc. → DOE, E.U.A.